

Ionic Liquid-based Electrolytes for Next Generation Batteries

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Germany

Beyond Lithium Ion, June 5, 2012



meet

Muenster Electrochemical Energy Technology

Research Groups:

- Prof. Martin Winter
- Prof. Stefano Passerini

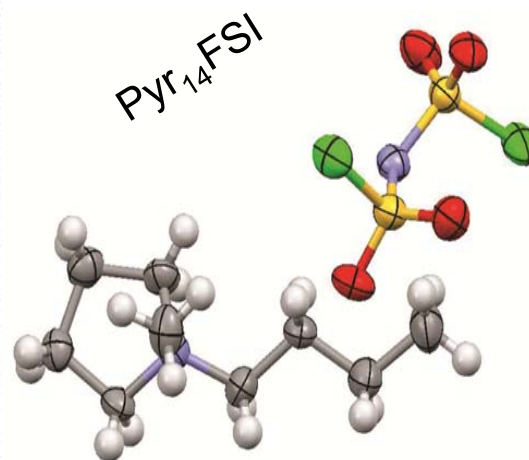
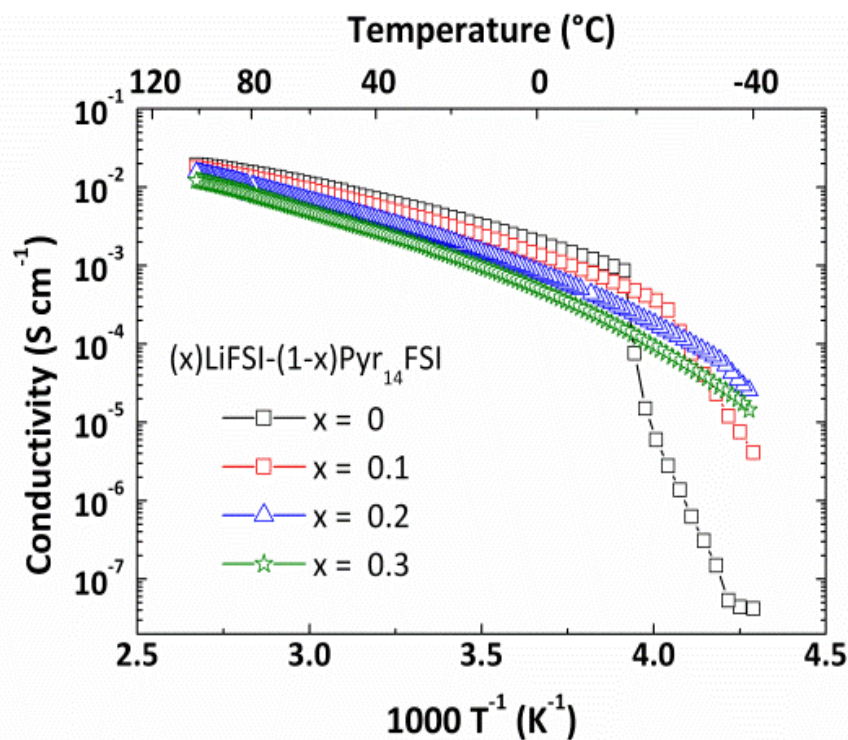
Young Researcher Groups:

- Dr. Alexandra Lex-Balducci (2009 -)
- Dr. Andrea Balducci (2009 -)
- Dr. Jie Li (2012 -)
- 50 Ph.D. Students
- 30 Scientists with Ph.D.
- 10 Bachelor Students



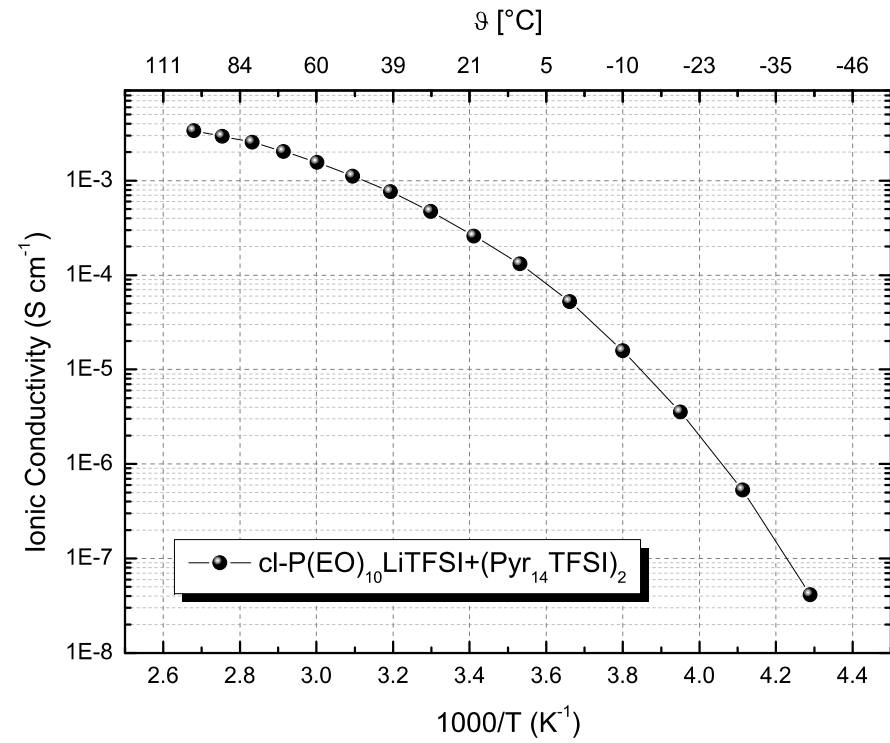
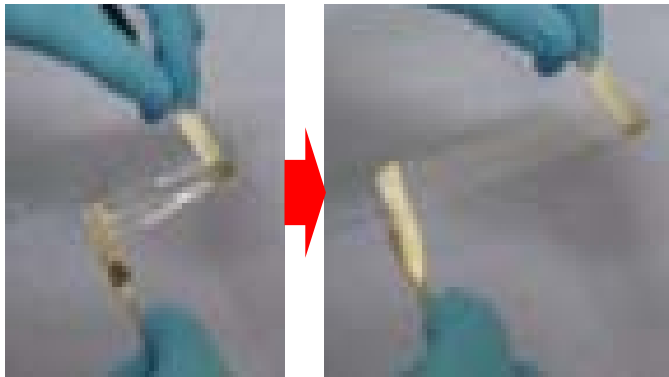
Ionic Conductivity in IL-based Materials for Energy Storage

Liquid electrolytes



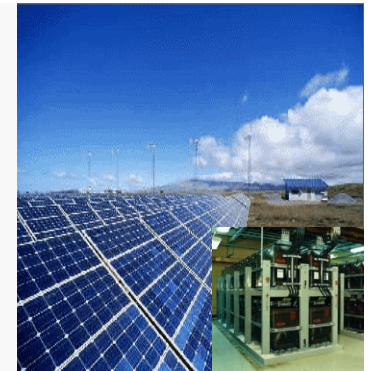
Ionic Conductivity in IL-based Materials for Energy Storage

Polymer electrolytes



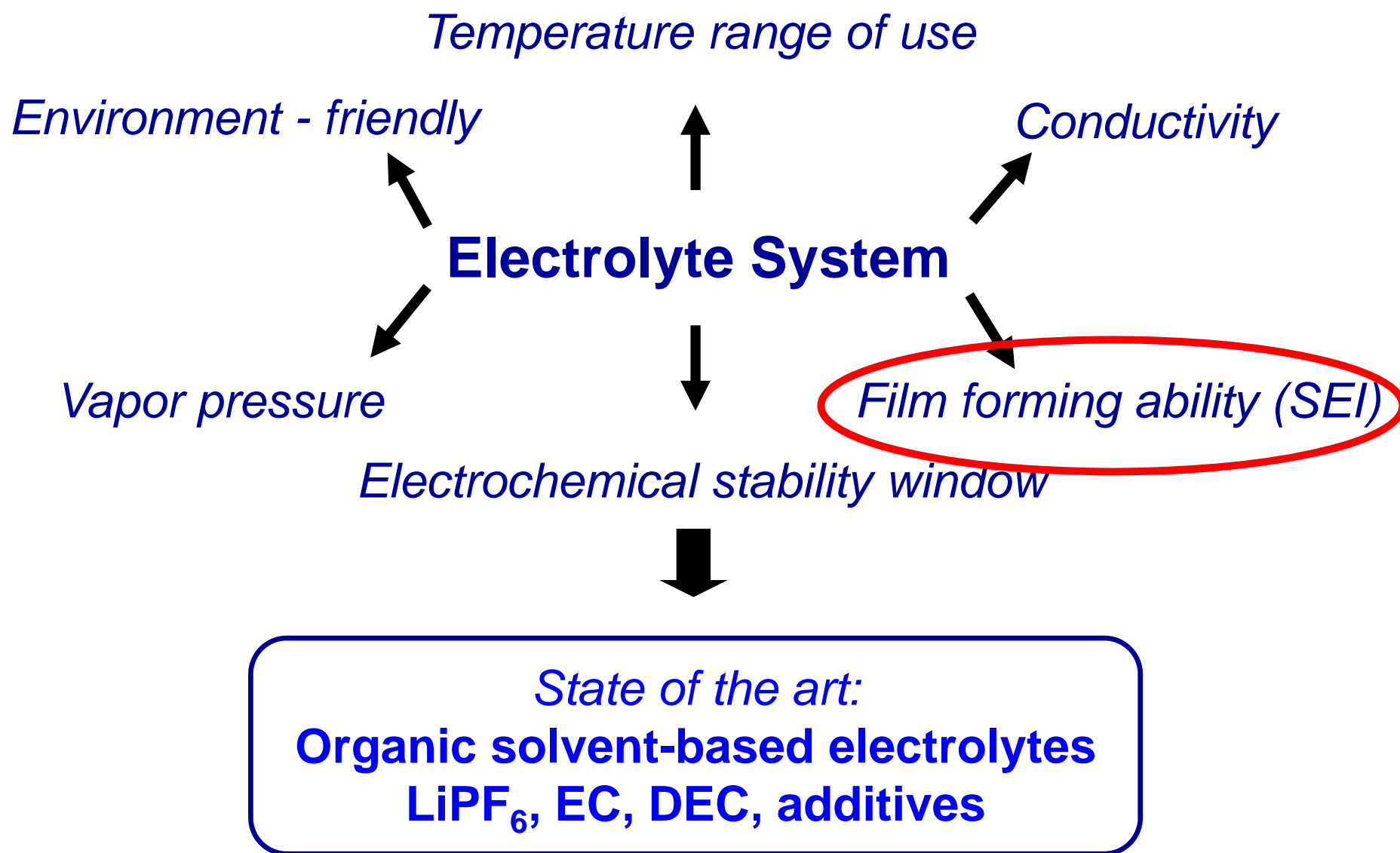
Li Ion Batteries Initiated a Technical Revolution....

... and might be ready for more

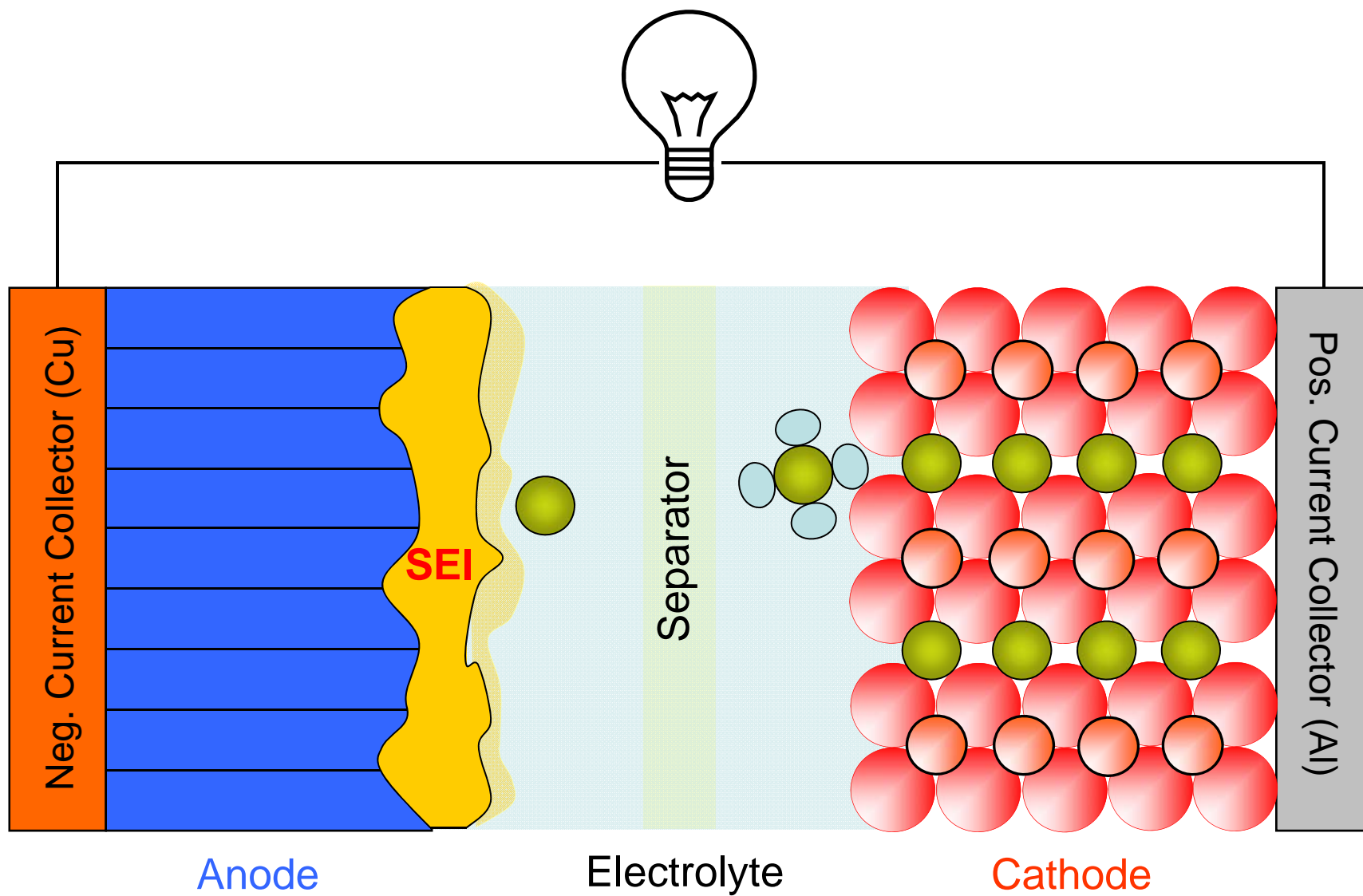


	Consumer Applications	LEV, Power Tools	Automotive HEV & EV	Stationary Elec. Storage
Market Size	€ 7Bio (2008)	medium € 0.7Bio (2008)	large € 25Bio (2020)	Very large
Market Introduction	1990 by Sony	2005	2012/15 (mostly HEV)	Realization has to be proven
Chances for Newcomers	Minimal	Moderate	(Very) Good	Depends on starting position
Typ. Battery-Size (kWh)	0.001 – 0.1	0.1 – 1	1 – 100	100 – 10.000

Key-role of Electrolyte in Li-Ion Batteries



Li-Ion cells: Role of the SEI film

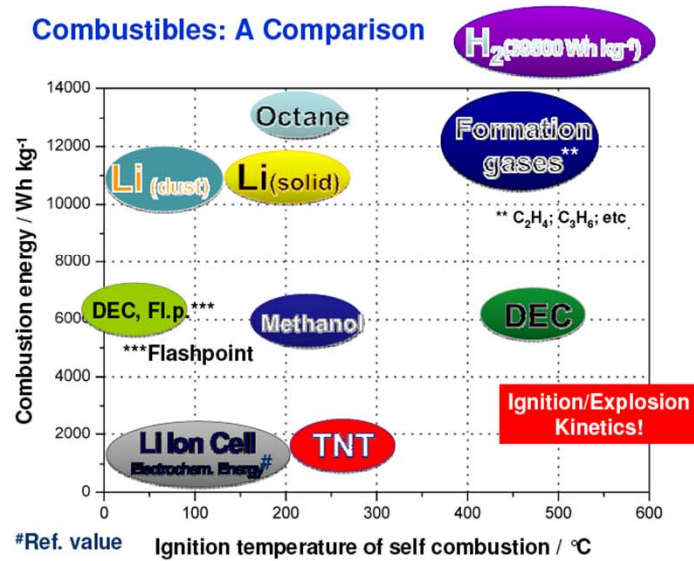


However

Safety of present Li-ion cells is still an open issue

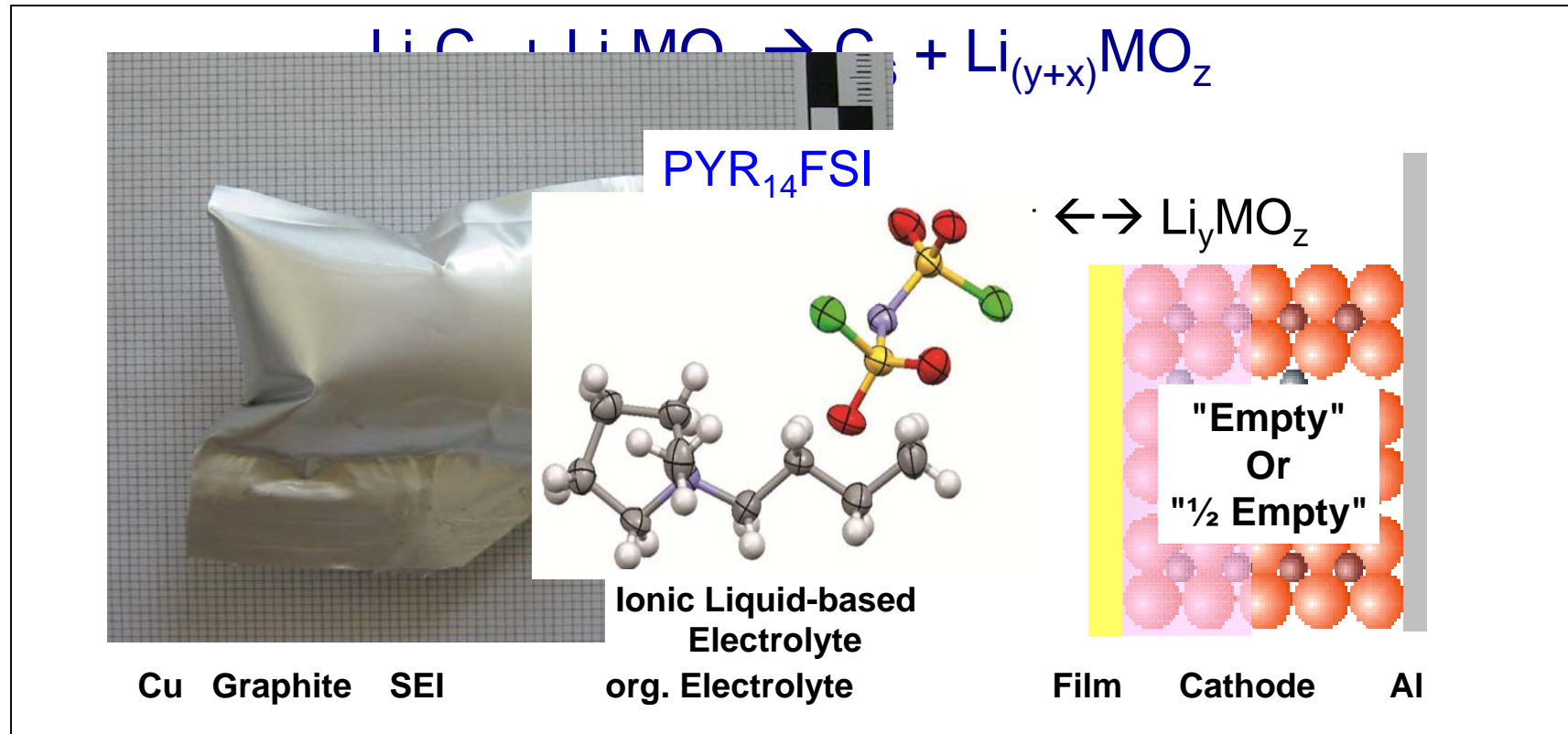


Combustibles: A Comparison



Towards Safer Batteries:

What do we learn from LIBs?



Key Step Forward: Replacement of volatile, flammable and electrochemically unstable electrolyte components

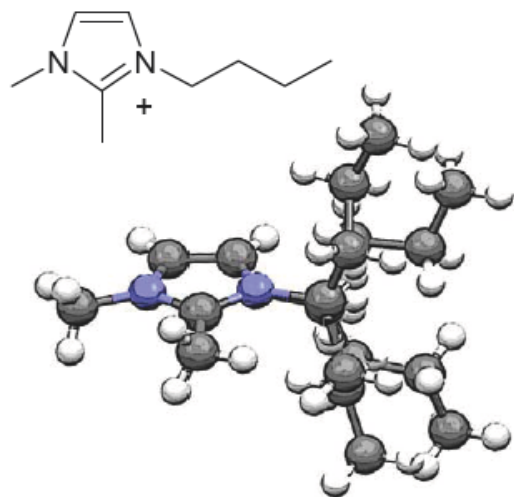
Room temperature molten salts (Ionic Liquids)

Ionic Liquids –Low Temperature Melting Salts

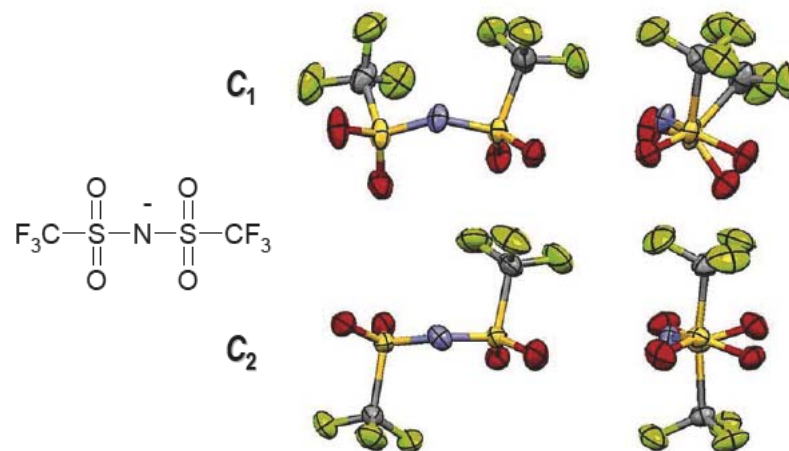
Salts, or mixtures of salts (composed solely of ions), which are liquid at low temperatures ($< 100^{\circ}\text{C}$) – often below RT

Weak interactions due to large cation and anion delocalization and low tendency to crystallize due to flexibility (anion) and dissymmetry (cation)

Ion Conformational Changes



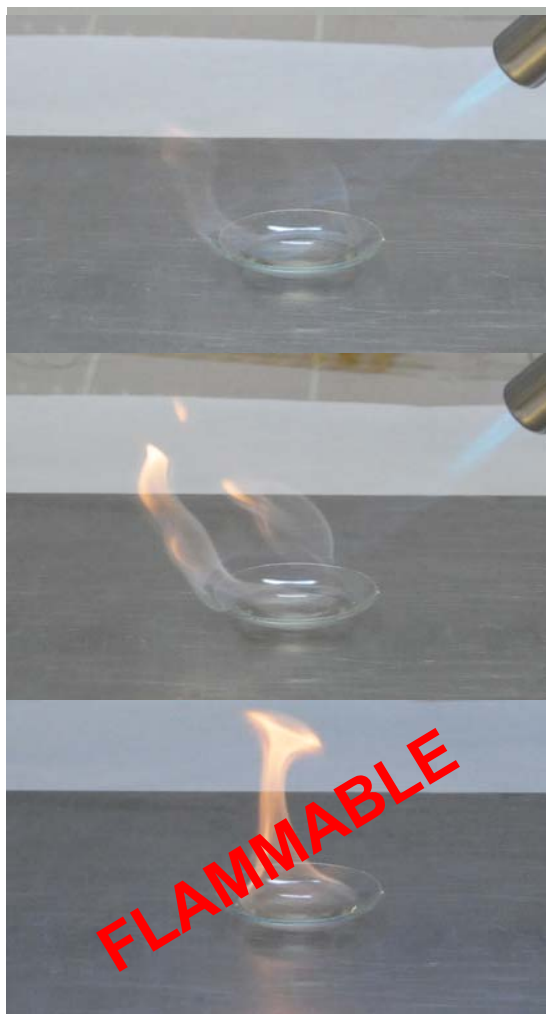
butyl conformations found in various crystal structures



well-know flexibility of the TFSI⁻ anion between 2 low-energy conformations

Flammability: Organic electrolyte vs. Ionic Liquids

ORGANIC ELECTROLYTE



0 sec

2 sec

3 sec

IONIC LIQUID ELECTROLYTE



NON-FLAMMABLE

Ionic Liquids (IL, RTIL, RTMS, etc.)

Properties

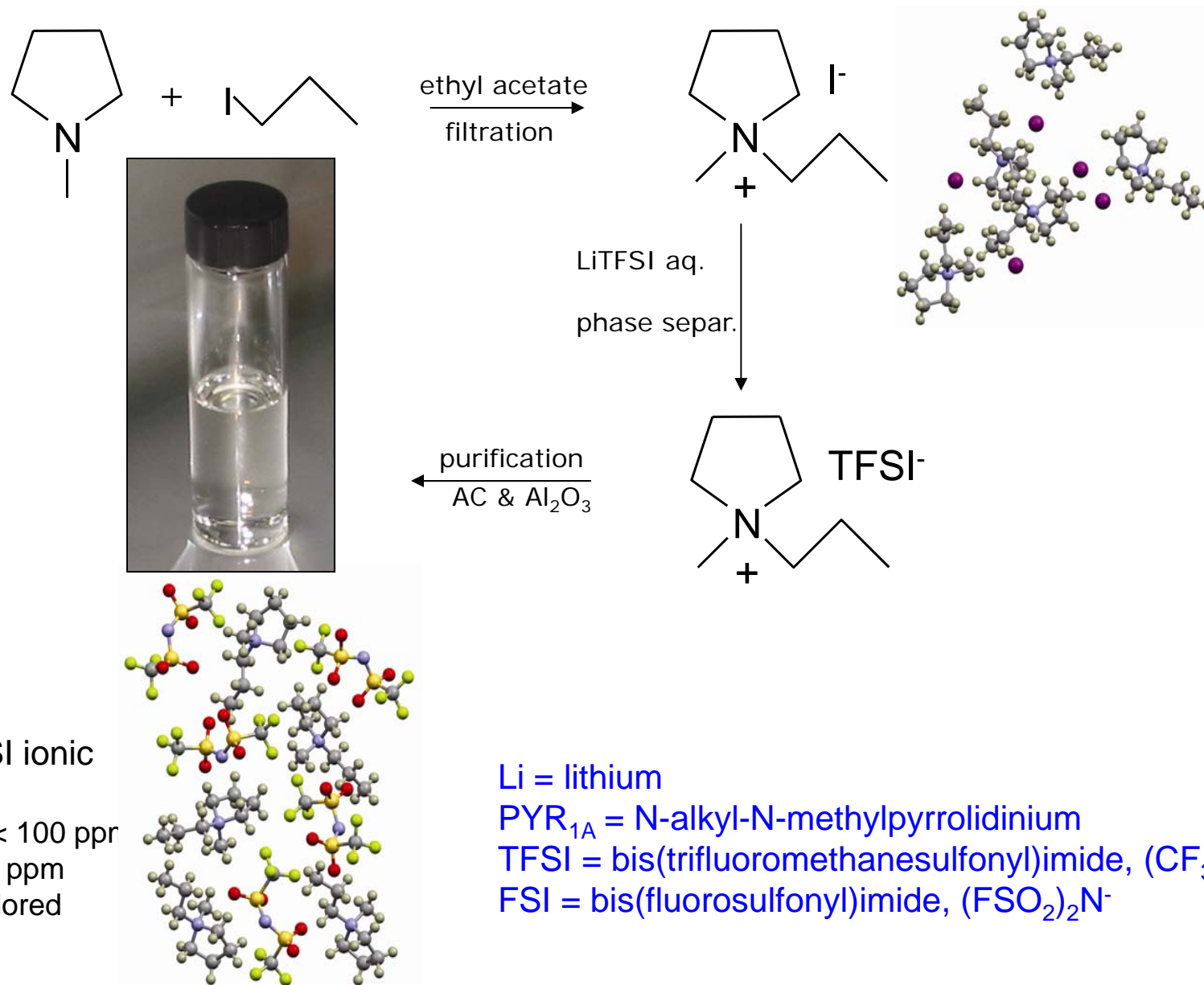
- ▶ Wide electrochemical window
- ▶ Wide liquid range
- ▶ Very low volatility at ambient pressure
- ▶ Good thermal stability
- ▶ High ionicity

Co-salts for
solid-state
polymer electrolytes

Solvents for
liquid electrolytes

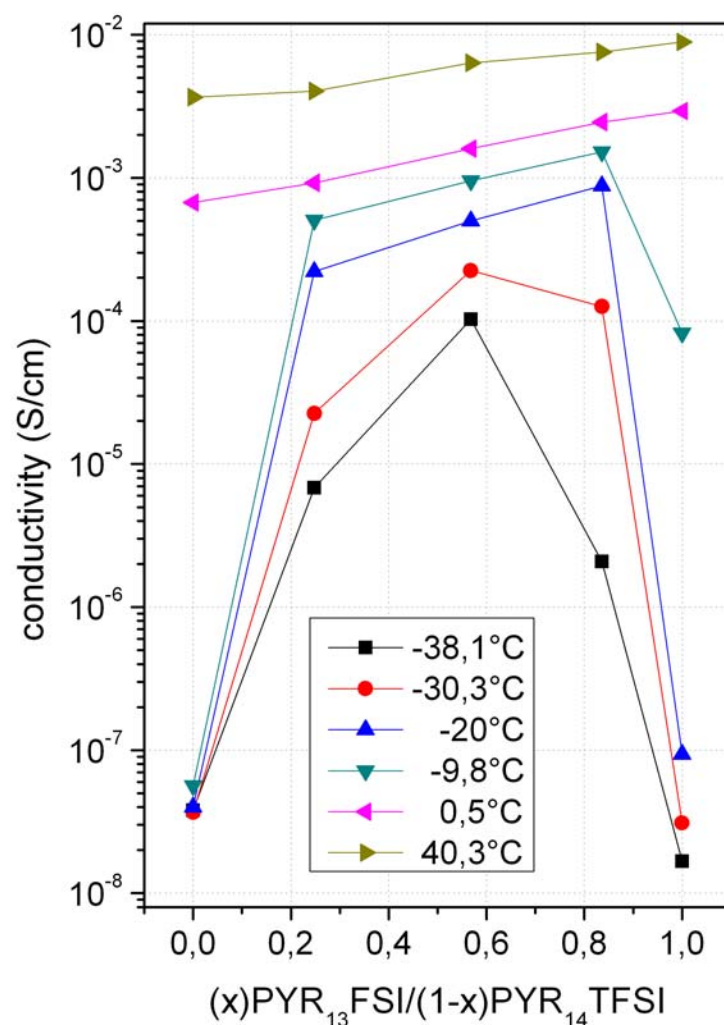
Thousands of possible ILs available

Synthesis of Ionic Liquids



Combining Ionic Liquids \longrightarrow Boosting Performance

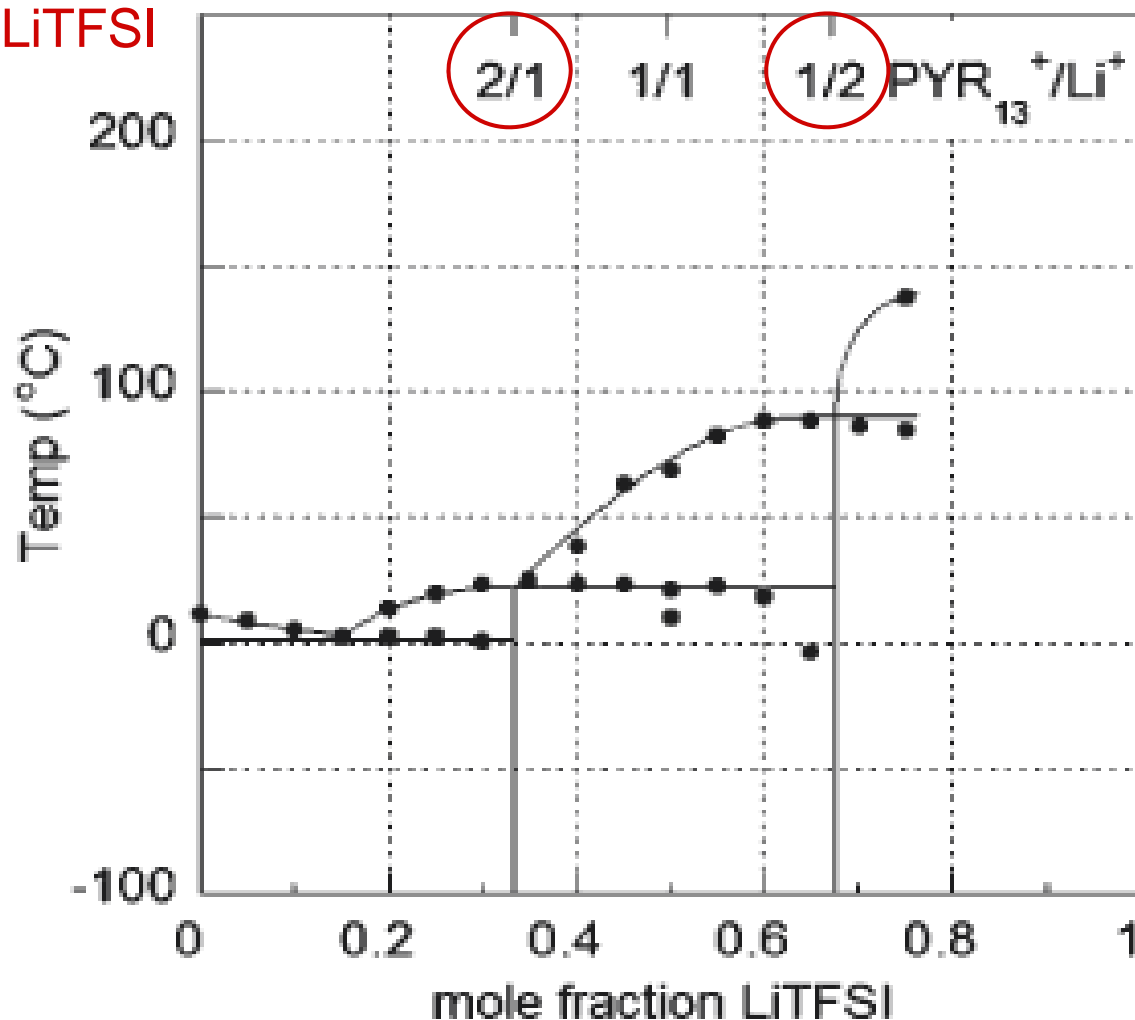
PYR₁₃FSI (m.p. -10° C) - PYR₁₄TFSI (m.p. -5° C)



Mixtures have much lower melting temperatures than constituent Ionic Liquids due to additional ion mismatching

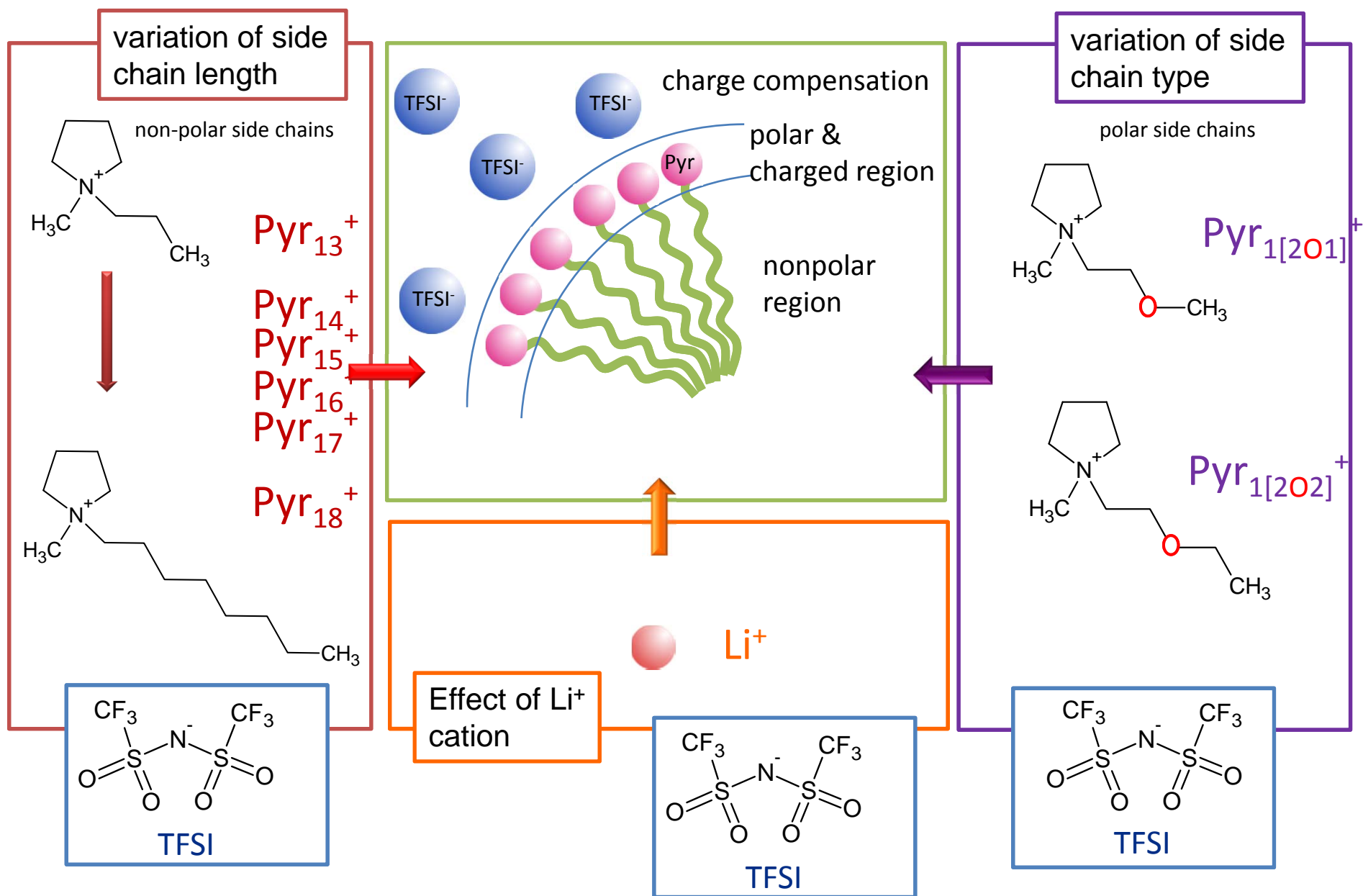
Are ionic liquids capable of dissolving Li salts?

PYR₁₃TFSI - LiTFSI



Yes, but only at low Li concentration because new crystalline phases are formed for high Li content

Aggregation of Pyr_{1N}TFSI ILs (NMR, Raman)



Local Motion in Ionic Liquids: NMR characterization (Two Step Model)

spin-lattice and spin-spin relaxation of a surfactant are sensitive to fast local motions and slow motions, respectively

$\omega\tau_s \gg 1$:

R_1 only influenced by fast bond rotations in the molecule that take place during the time τ_f – not influenced by aggregation of molecules

R_2 is largely influenced by the slow reorientations and experiences the molecular aggregation

spectral density:

$$J(\omega) = S^2 \frac{2\tau_s}{1 + \omega^2 \tau_s^2} + (1 - S^2) \frac{2\tau_f}{1 + \omega^2 \tau_f^2}$$

$$S = \frac{(3\cos^2(\theta) - 1)}{2}$$

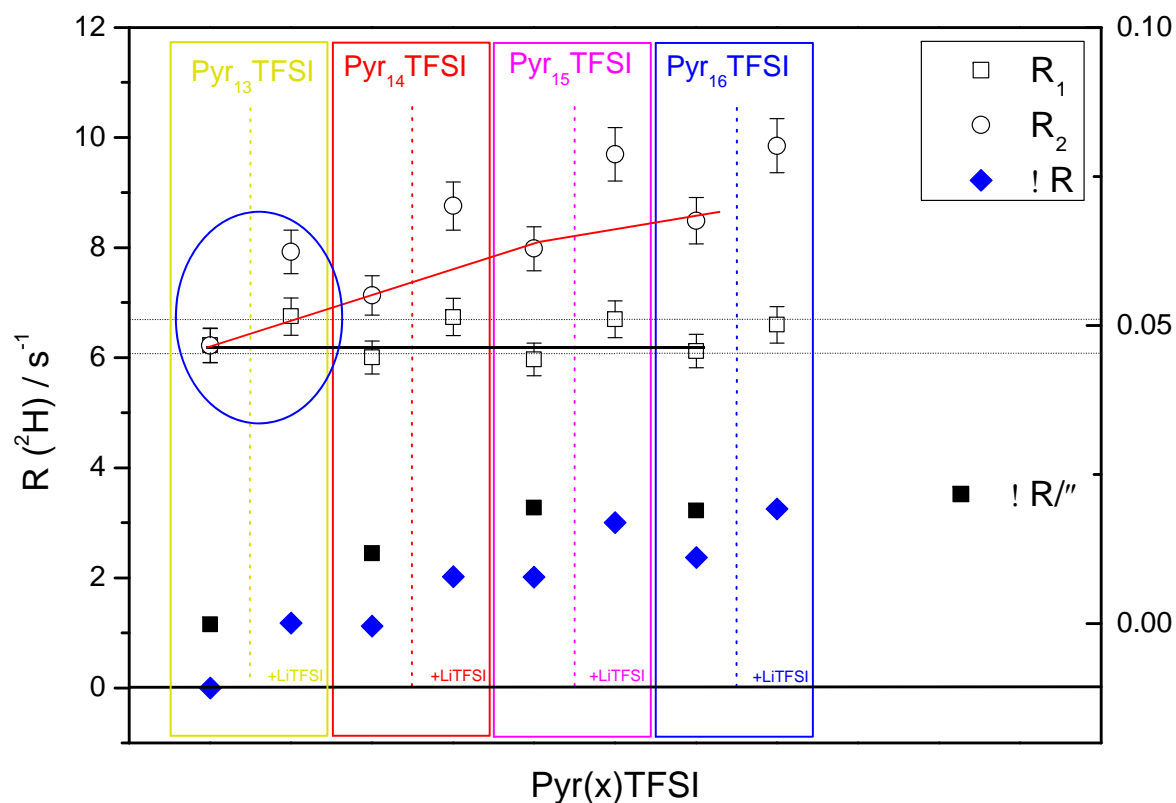
τ_f characterizes the fast internal motions (bond rotations)

τ_s describes rotational tumbling of the molecule

$$\Delta R = R_2 - R_1 \cong \frac{9\pi^2}{20} \chi^2 S^2 \tau_s$$

Aggregation in Pyr_{1x}TFSI

NMR measurements: T1 and T2 vs x (in Pyr_{1x}TFSI)



R_1 independent on chain length
 \Rightarrow fast local motions not influenced

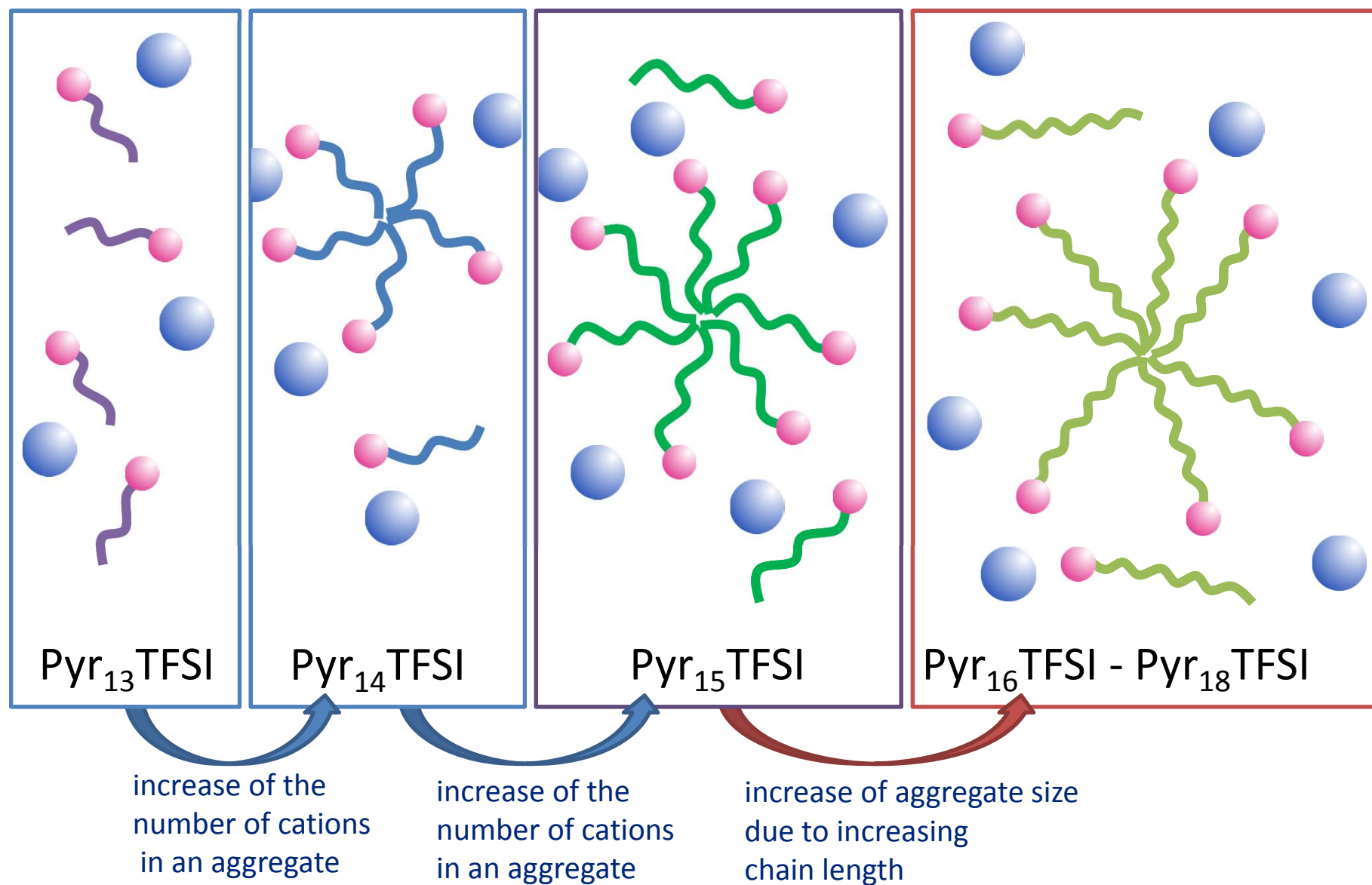
R_2 dependent on chain length
 \Rightarrow slow local motions influenced

two different linear dependences
 \Rightarrow different aggregation mechanisms

In presence of Li⁺ ΔR differs from 0
 already for propyl chain
 \Rightarrow Li forces cation aggregation

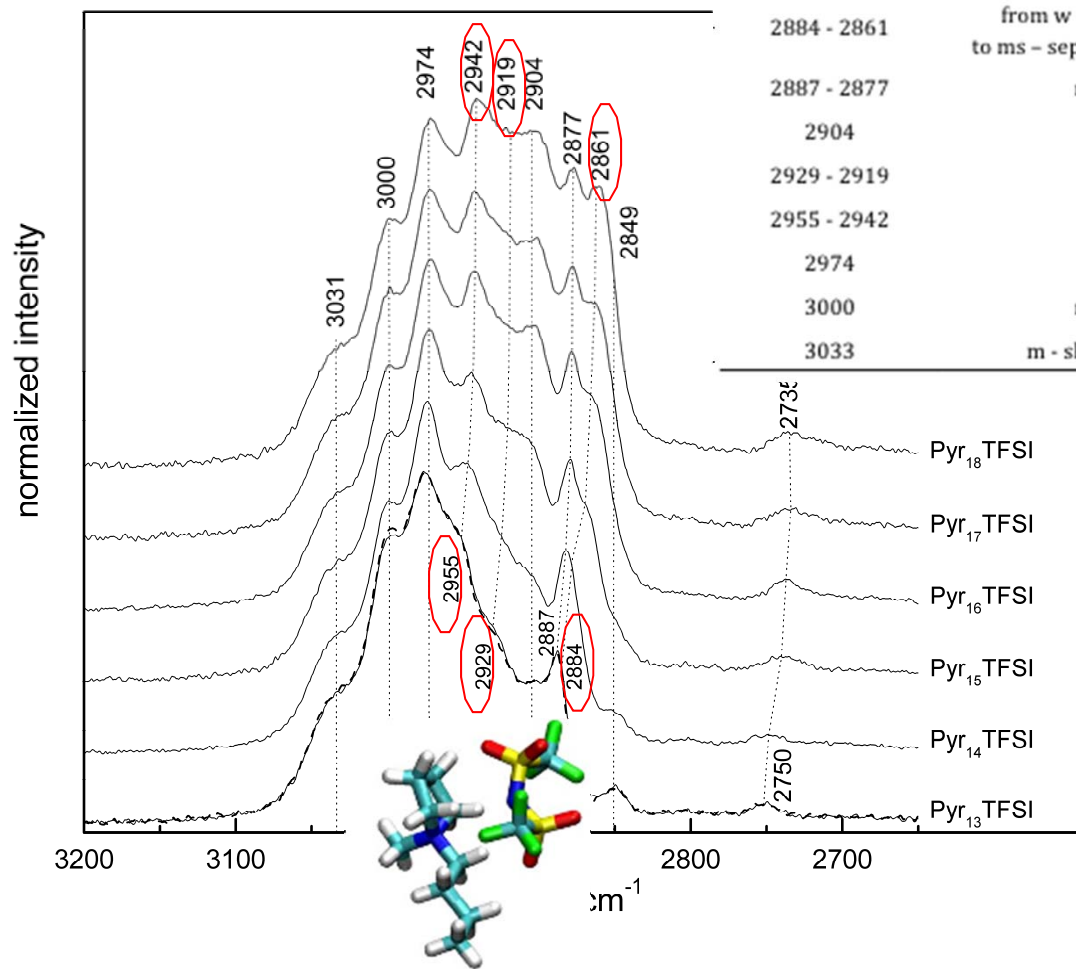
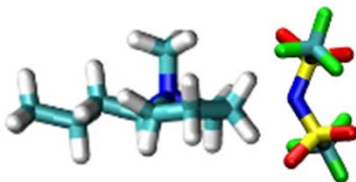
Typical behavior of micelles indicating the formation of aggregates

Aggregation type?



Aggregation in Pyr_{1x}TFSI

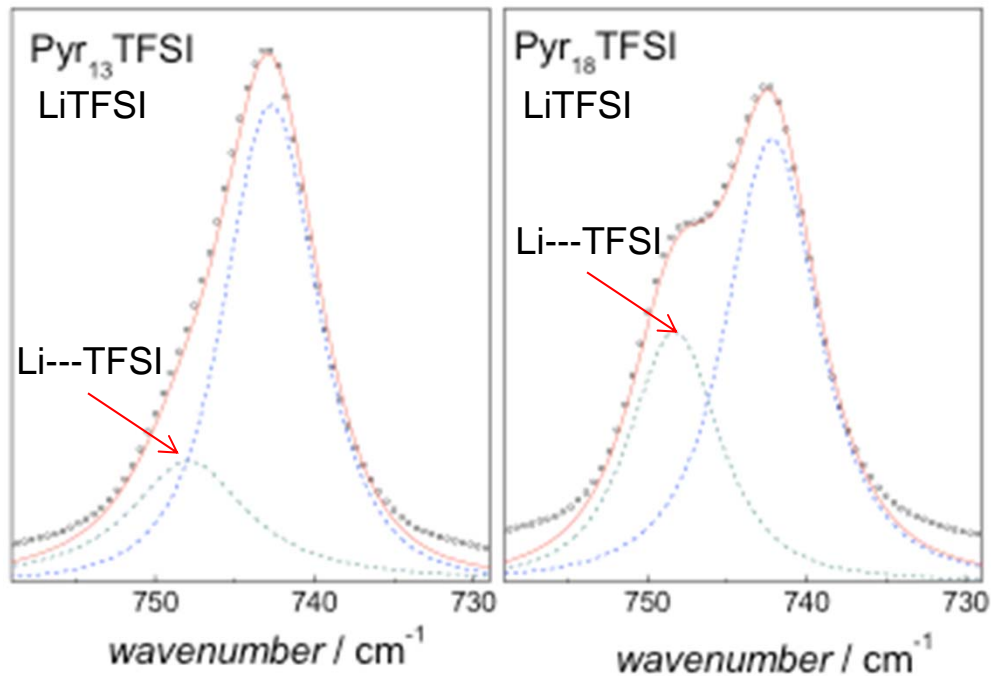
Raman measurements



position / cm ⁻¹	relative intensity	assignment
2750 - 2736	w	N ⁺ -C _x shifting to lower wavenumbers with increasing side chain length x
2849	w	
2884 - 2861	from w - shoulder to ms - separated mode	
2887 - 2877	ms	-CH ₂ - in ring
2904	w	alkyl chain
2929 - 2919	w	-CH ₂ - in ring
2955 - 2942	m	-CH ₂ - in ring
2974	s	alkyl chain
3000	ms	ν ₃ (C-H) in C-CH ₃
3033	m - shoulder	ν ₄ (C-H) in N ⁺ -CH ₃

The aggregation is confirmed by the shift of characteristic Raman peaks

Cation aggregation: Li⁺ --- TFSI⁻ ion-pairing



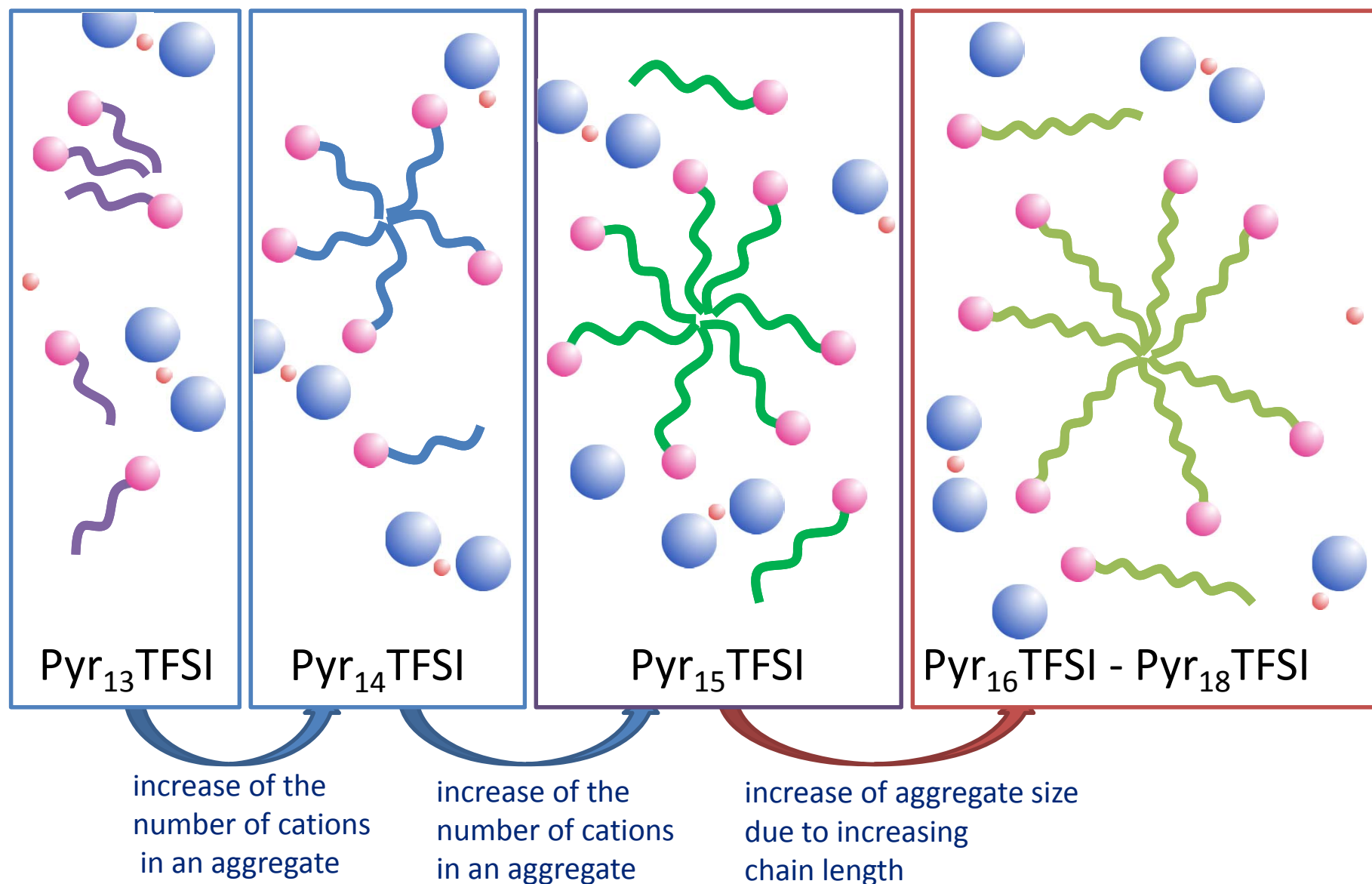
@ 742 cm⁻¹ expansion and contraction of the whole TFSI-anion
 @ 748 cm⁻¹ damped TFSI-anion expansion and contraction due to Li⁺ clustering

increase of mode @ 748 cm⁻¹ with increasing side chain length
 ⇒ more anions are clustering the Li-cation when more Pyr-cations cluster with themselves

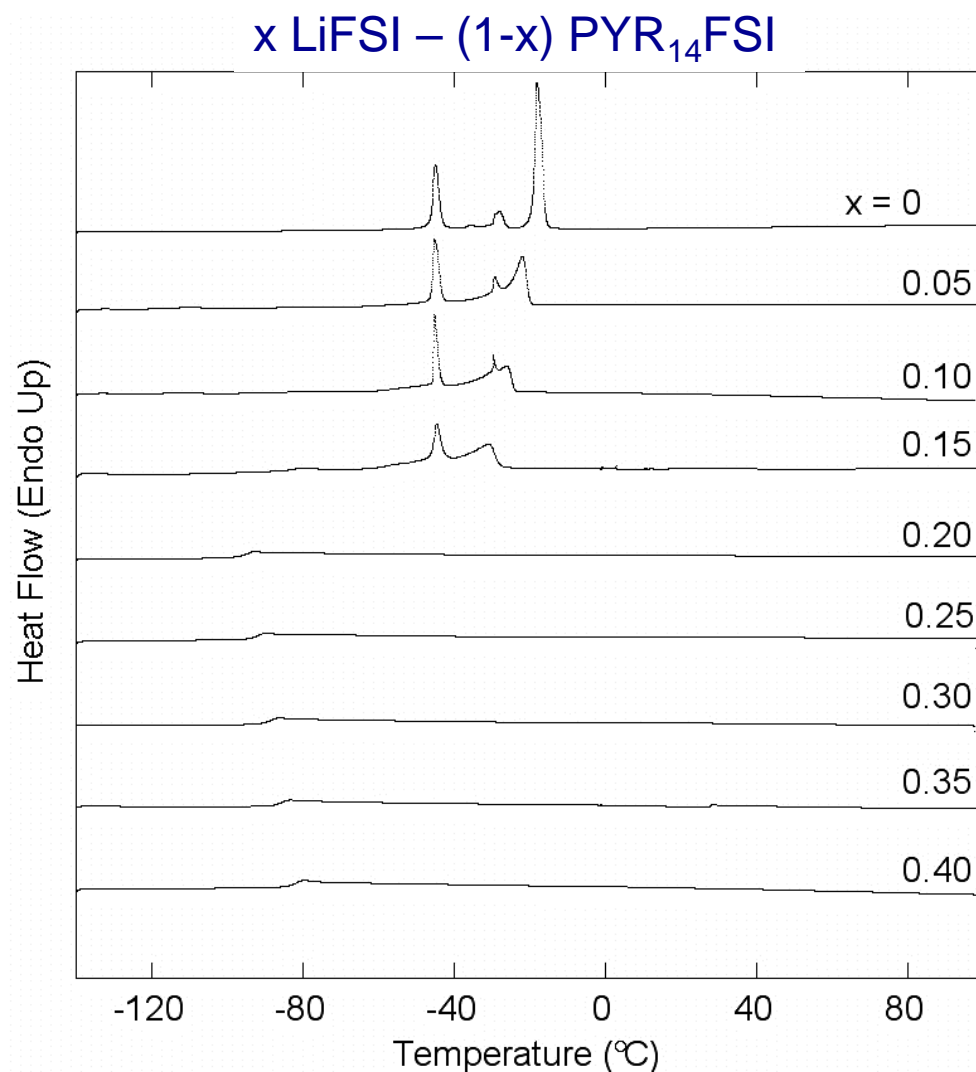
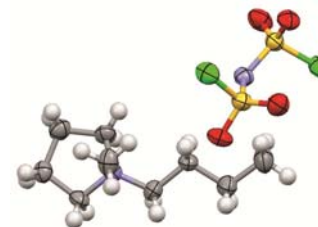
x in Pyr _x TFSI	[Pyr _x TFSI] _{0.85} [LiTFSI] _{0.15}		
	free TFSI position / cm ⁻¹	Li ⁺ ...TFSI position / cm ⁻¹	$n_{in Li^+ \dots (TFSI^-)_n}$
3	742.7	747.9	1.90±0,01
4	742.3	748.3	2.12±0,01
5	742.4	748.1	2.34±0,01
6	742.2	748.3	2.33±0,02
7	742.2	748.3	2.34±0,02
8	742.2	748.3	2.33±0,02

increase of anions,
 which surround the Li⁺

PYR^+_{1x} aggregate formation in presence of Li^+



Tailored electrolyte: Li salt in PYR₁₄FSI

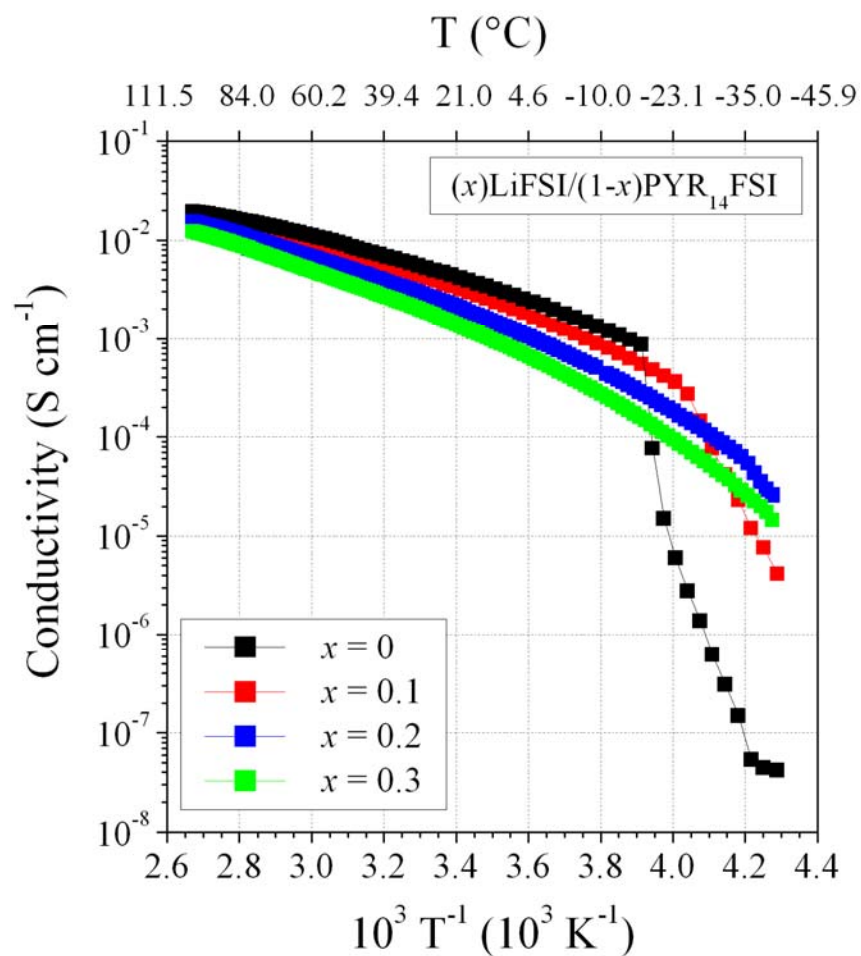


**Sub-ambient temperature
melting electrolytes**

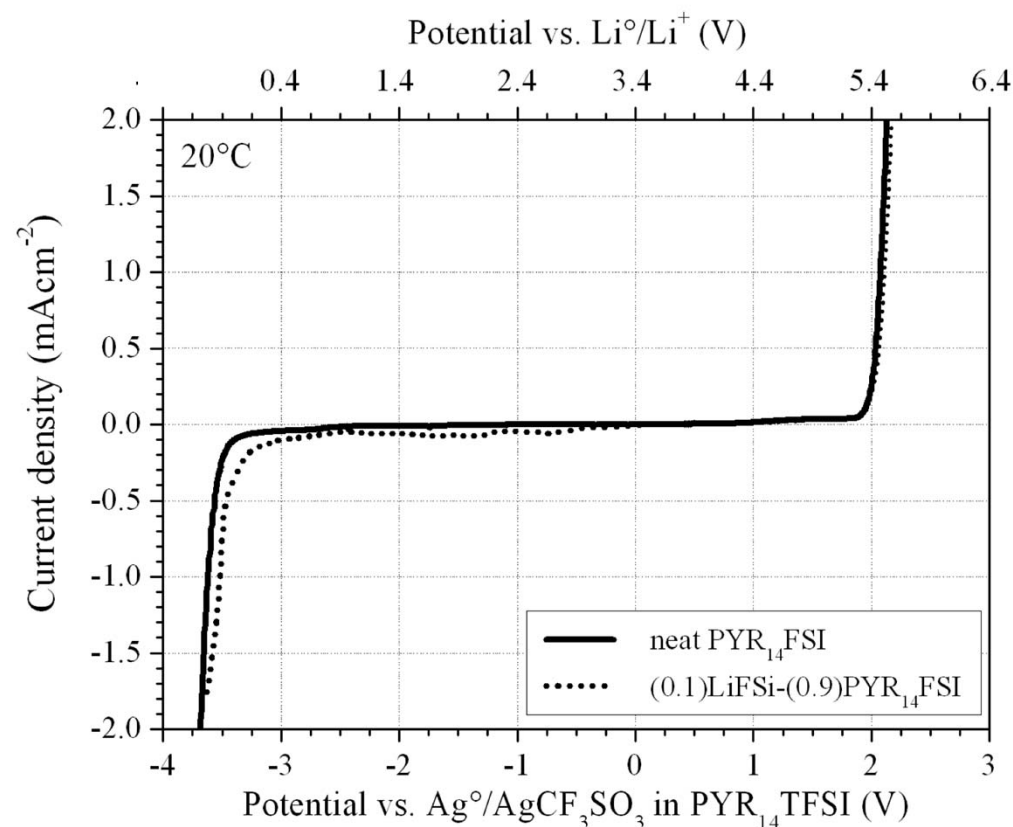
**The electrolytic mixture
0.85 PY₁₄FSI - 0.15 LiFSI
melts below -40° C**

**The electrolytic mixtures with
 $x > 0.15$ show only glass
transition below -60° C**

LiFSI – PYR₁₄FSI mixtures for electrolytes

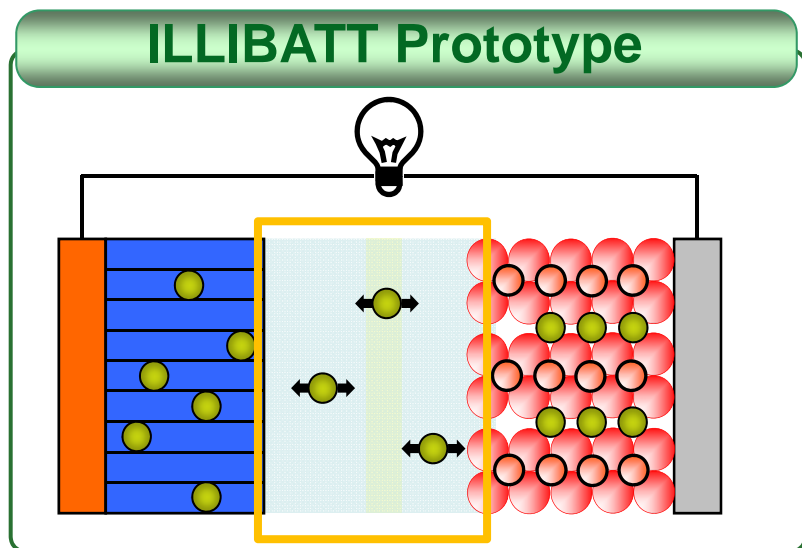


High ionic conductivity at sub-ambient temperatures



Wide Electrochemical Stability Window (ESW)

Ionic Liquid-based Lithium-ion Batteries

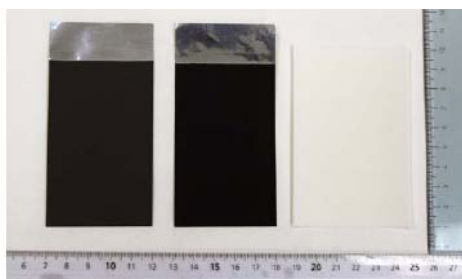


ANODE: $\text{Li}_4\text{Ti}_5\text{O}_{12}$ (LTO)

CATHODE: LiFePO_4 (LFP)

BINDER: CMC

ELECTROLYTE: 0.1LiTFSI – 0.9 $\text{PYR}_{14}\text{FSI}$



ELECTRODES



STACK OF ELECTRODES



ELECTROLYTE



PROTOTYPE

Active material loading: **0.6 – 0.8 mAh/cm²**

Total Capacity: **0.7 – 0.8 Ah**

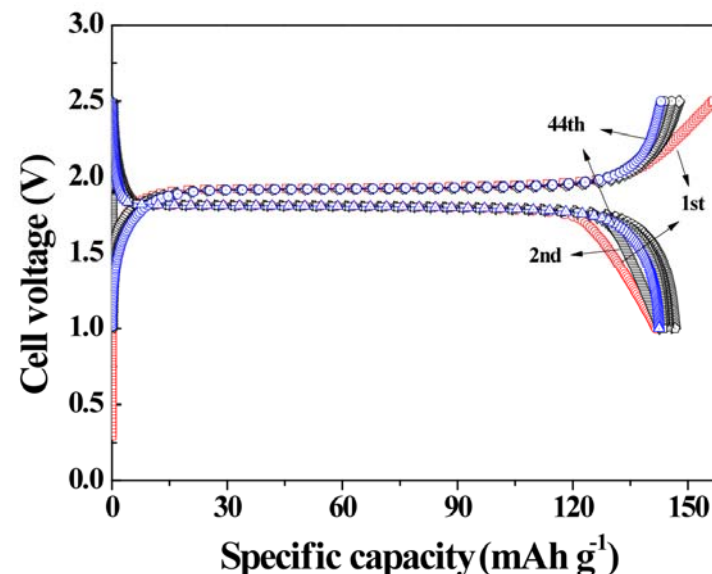
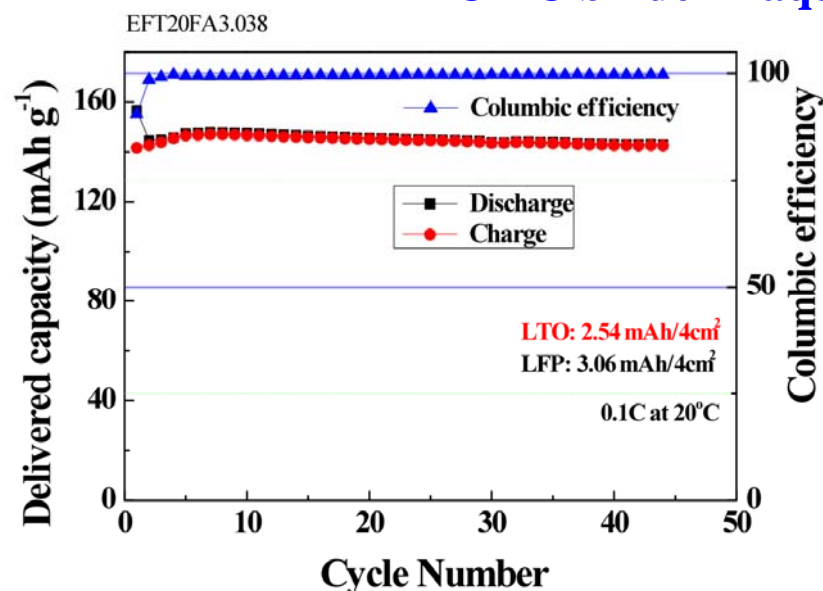
Stack (12 layers) of bipolar electrodes

Li-Ion cells (LFP-LTO) with Ionic Liquid-based Electrolytes

LTO / 0.9PYR₁₄FSI – 0.1LiTFSI / LFP

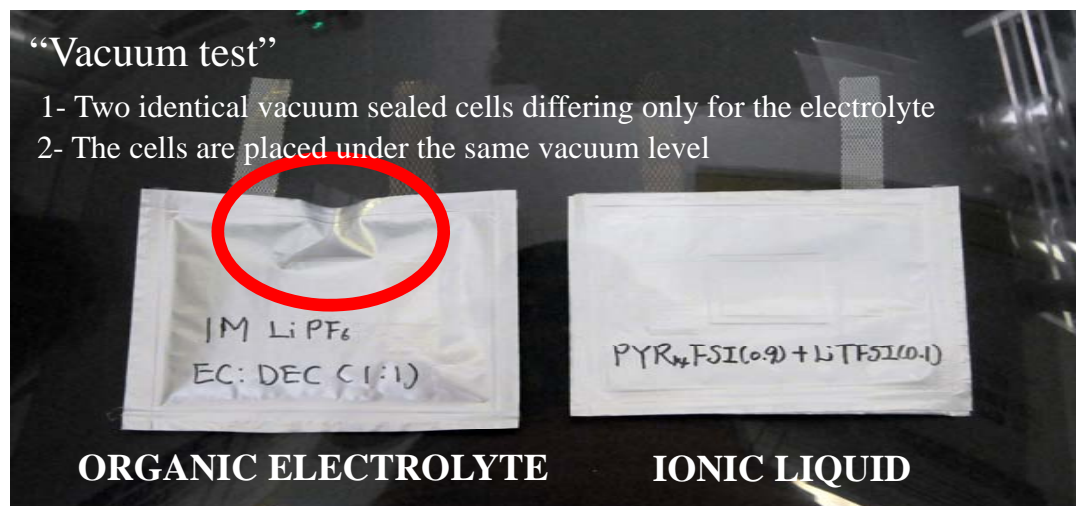
Pouch Cell, glass fiber, 20° C

CMC binder – aqueous slurries – Al collectors



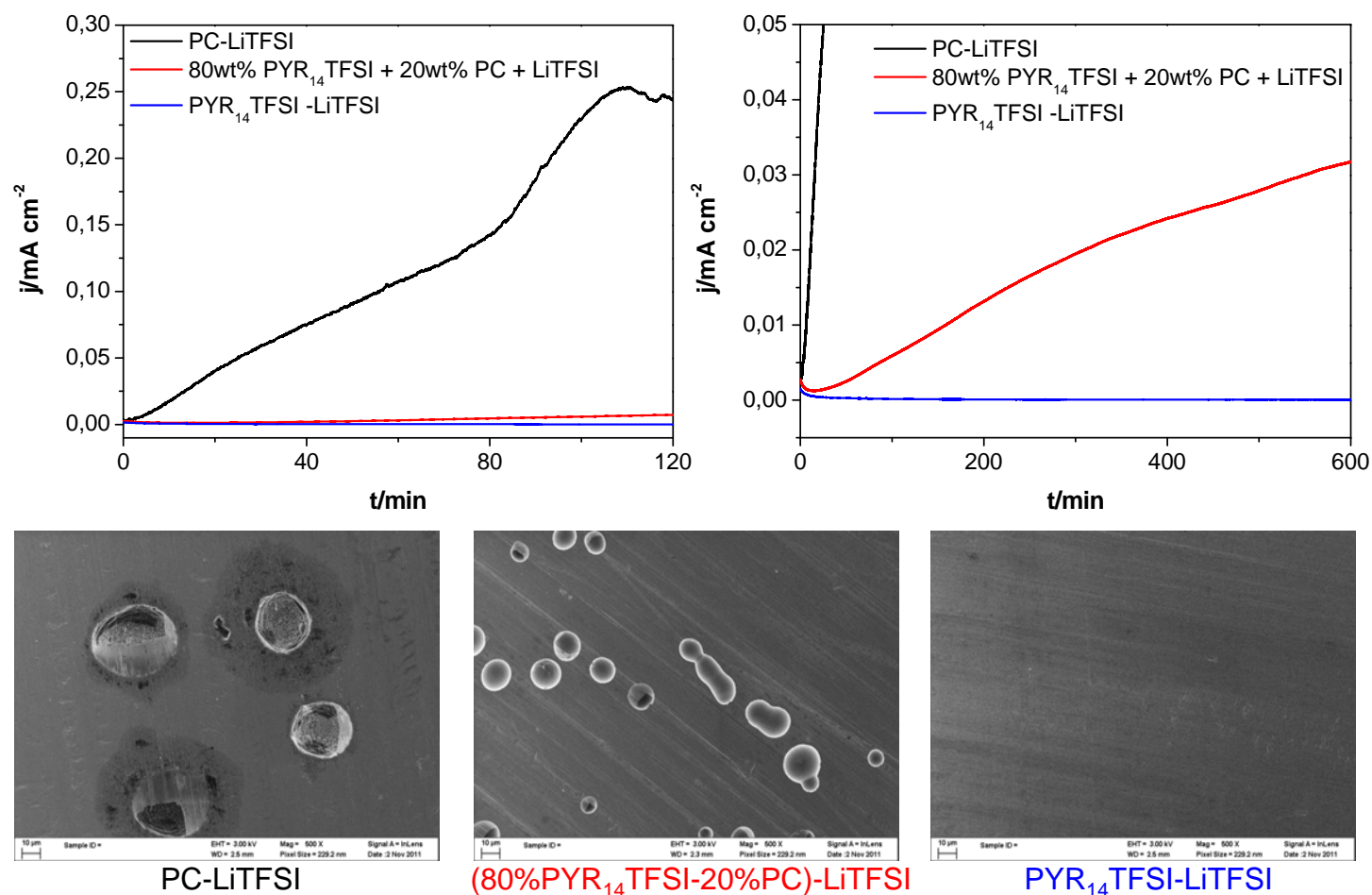
Very promising
cycling performance
Operation under
vacuum

Appetecchi G. B, et al.
J. POWER SOURCES 196, 6703-6709, 2011



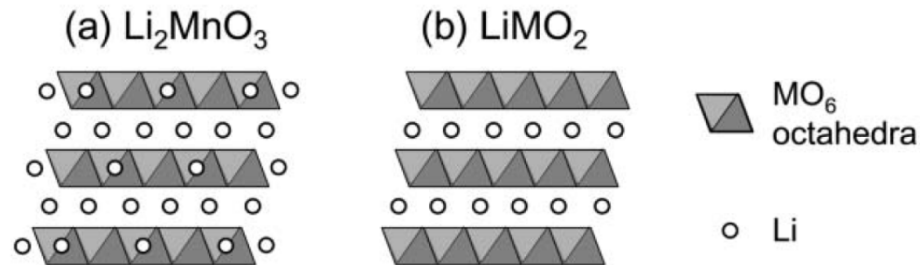
Aluminum in $\text{PYR}_{14}\text{TFSI}$ - LiTFSI electrolytes

Chronopotentiometry at 4.6 V vs. Li/Li

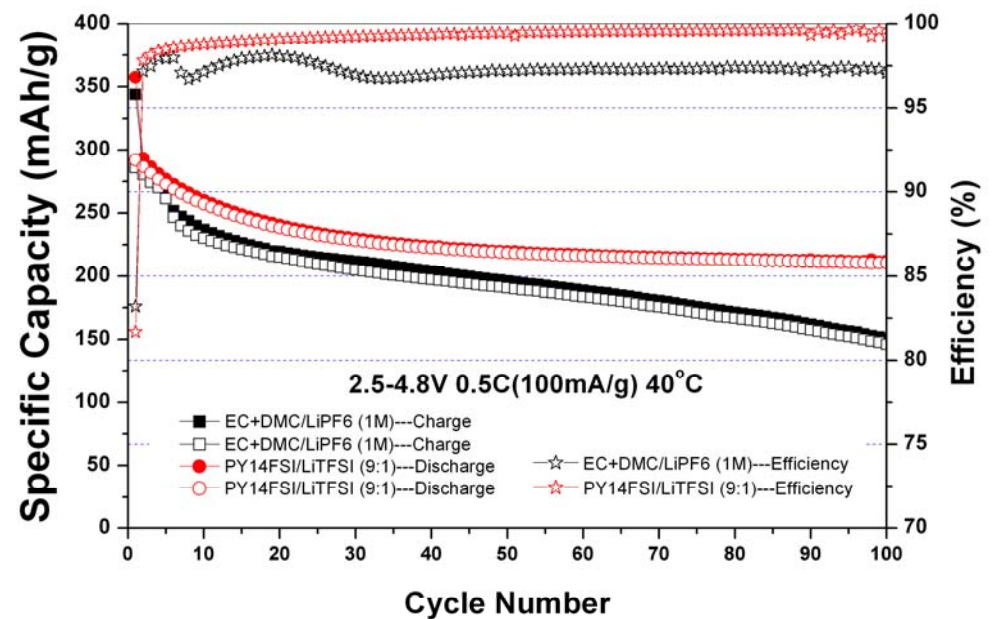
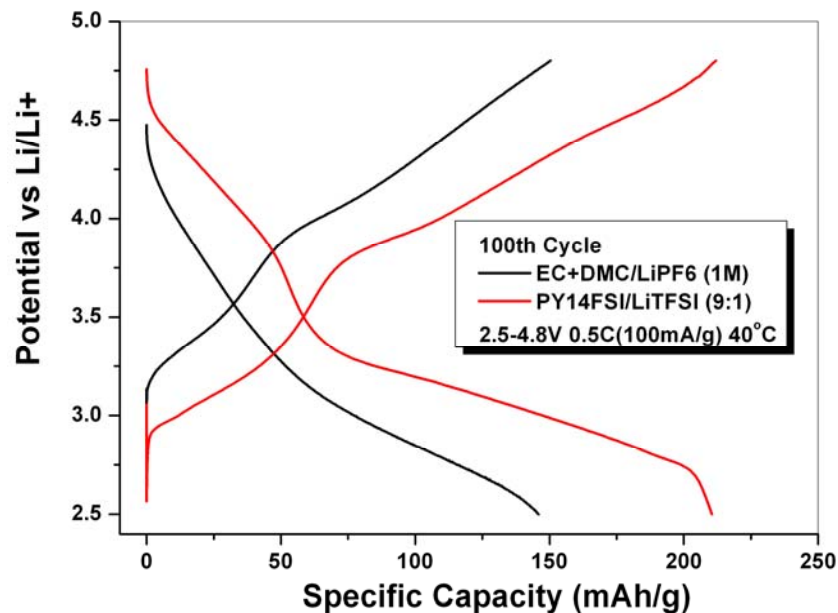


Al corrosion suppressed with TFSI-based electrolytes

High Voltage Electrodes: IL vs Carbonates @ 4.8 V



Li / Electrolyte / $\text{Li}[\text{Li}_{0.2}\text{Mn}_{0.56}\text{Ni}_{0.16}\text{Co}_{0.08}]\text{O}_2$
Pouch Cell @ 40° C and C/2



Very high capacity after long-term cycling
High coulombic efficiency (pract. 100%)

High-Energy (Next-Generation) Battery Technologies



Pb-acid 3000 kg
Ni-MH 1200 kg

Super- Battery < 200kg

>500 Wh/kg

**Revolutionary
Technology-
Change**

250 Wh/kg*

**Estimated
limit of
Lithium-Ion
Technology**

400 kg

Li-ion Batteries

170 Wh/kg*

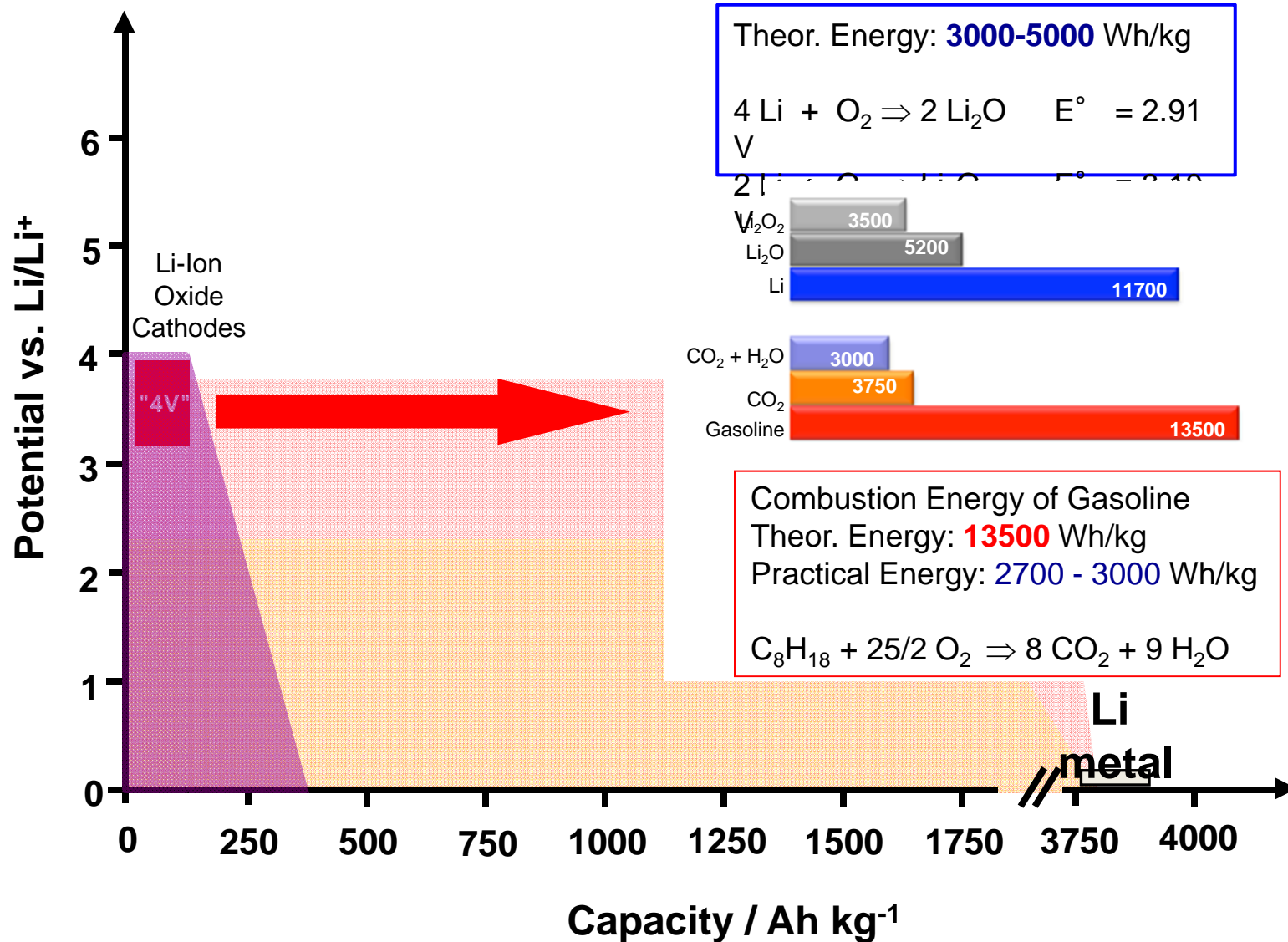
600 Kg

Present

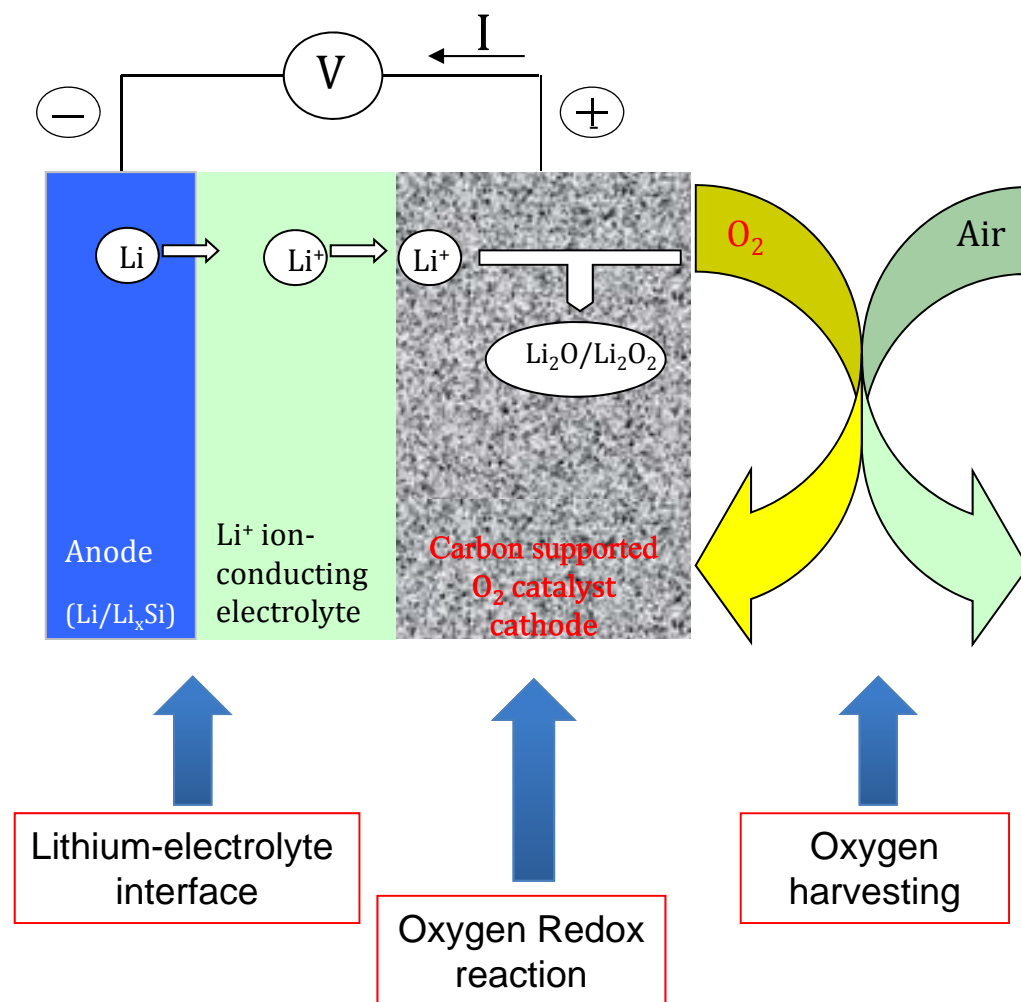
2017

Year

Li-Element batteries



Li-Air batteries: Splitting the Issues



EU Project LABOHR

FP7-2010-GC-ELECTROCHEMICAL STORAGE
265971

Target:
500Wh/Kg & 200W/kg at the
battery pack

Participant organisation name	Country
Westfaelische Wilhelms-Universitaet Muenster (WWU)	DE
Tel Aviv University (TAU)	IL
Consejo Superior de Investigaciones Cientificas (CSIC)	ES
Kiev National University of Technology and Design (KNUTD)	UKR
University of Bologna (UNIBO)	IT
University of Southampton (SOTON)	UK
SAES Getters S.p.A. (SAES)	IT
Chemetall	DE
AVL List GmbH (AVL)	AT
Volkswagen (VW)	DE
European Research Services GmbH (ERS)	DE

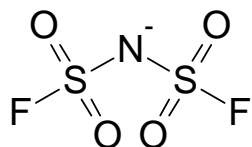
KEY ISSUES OPEN

Conventional electrolytes based on carbonates and ethers have been ruled out because of insufficient stability for ORR

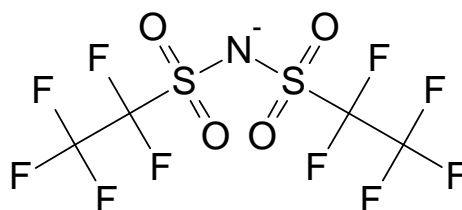
Investigated ILs for Li-Air cells

4 anions vs. 2 cations

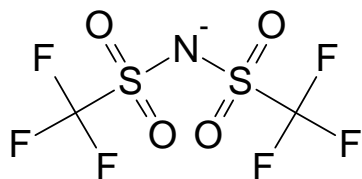
FSI



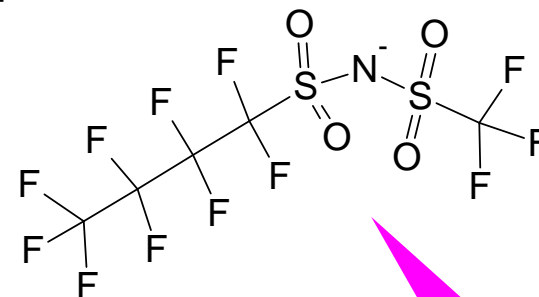
BETI



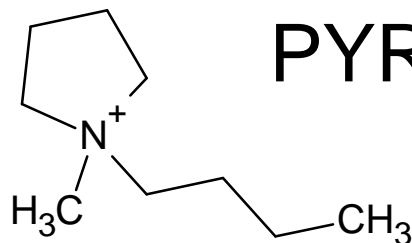
TFSI



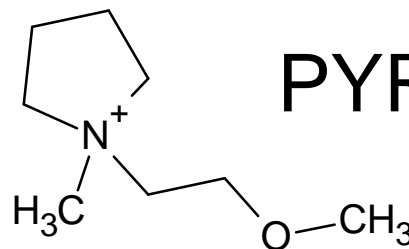
IM14



PYR₁₄⁺

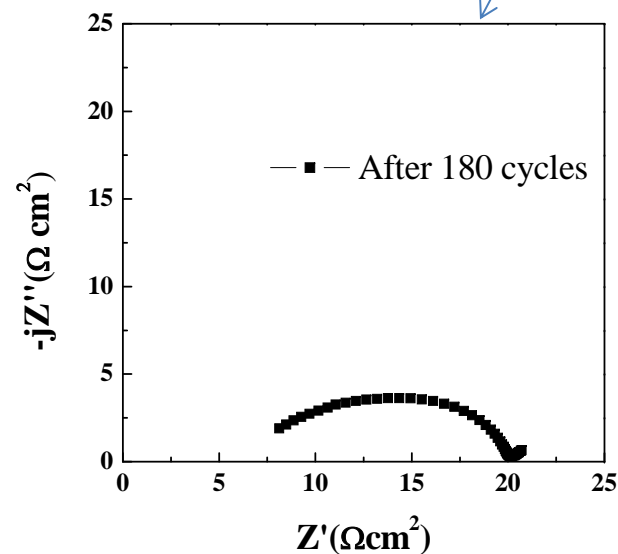
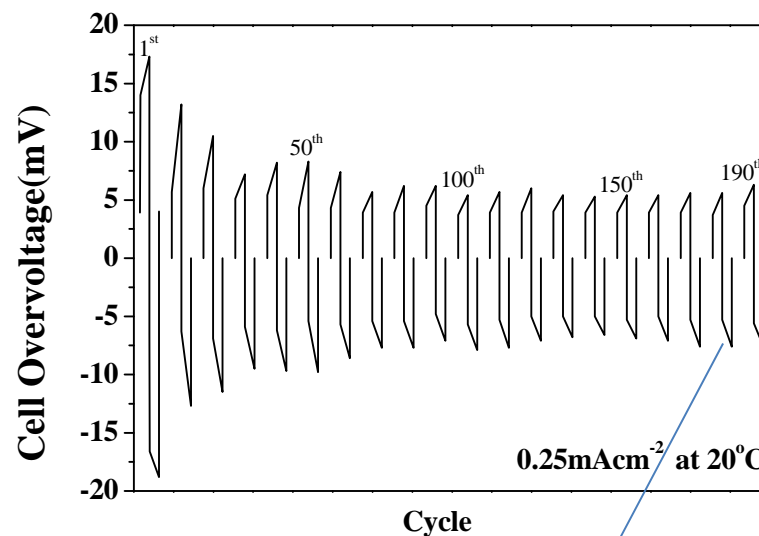
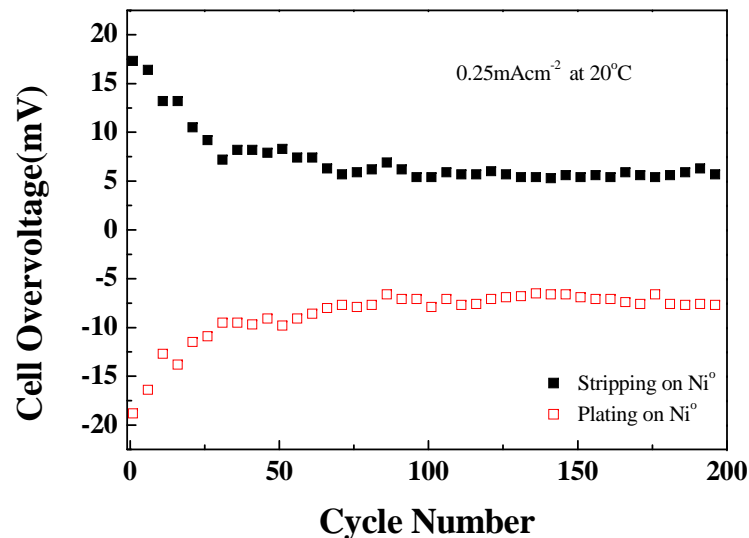


PYR_{12O1}⁺



Li plating/stripping from IL-based electrolytes

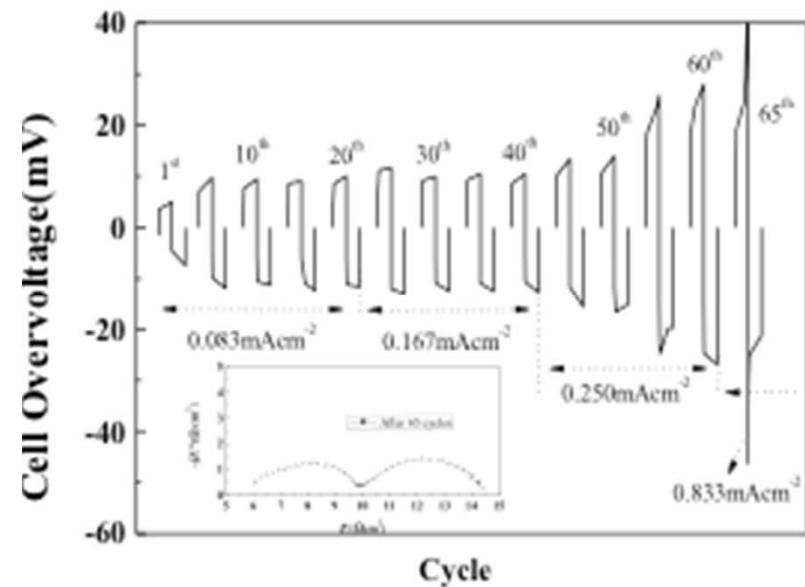
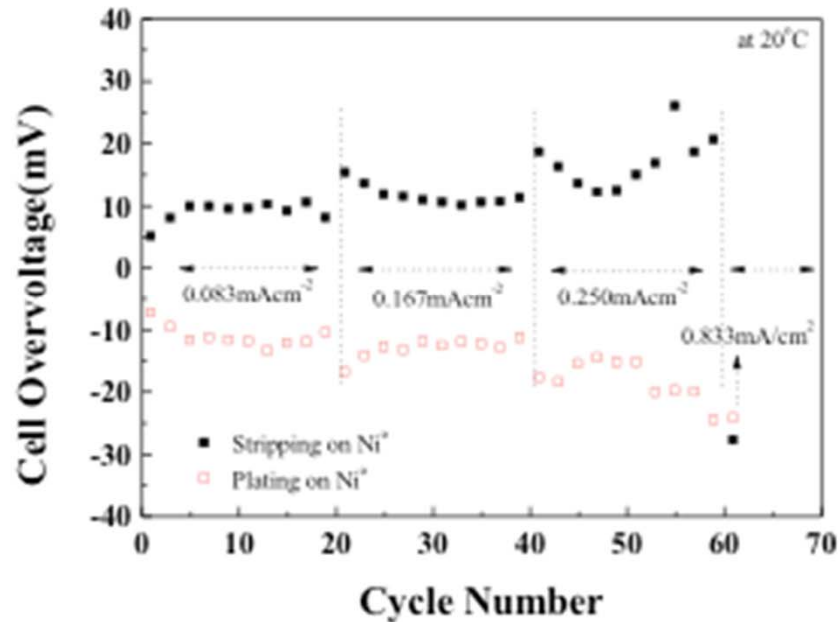
Li / 0.1LiTFSI-0.9PYR₁₄FSI/Ni (20° C, 0.25 mAcm⁻², sealed cell)



Very promising cycle performance

Li plating/stripping from IL-based electrolytes

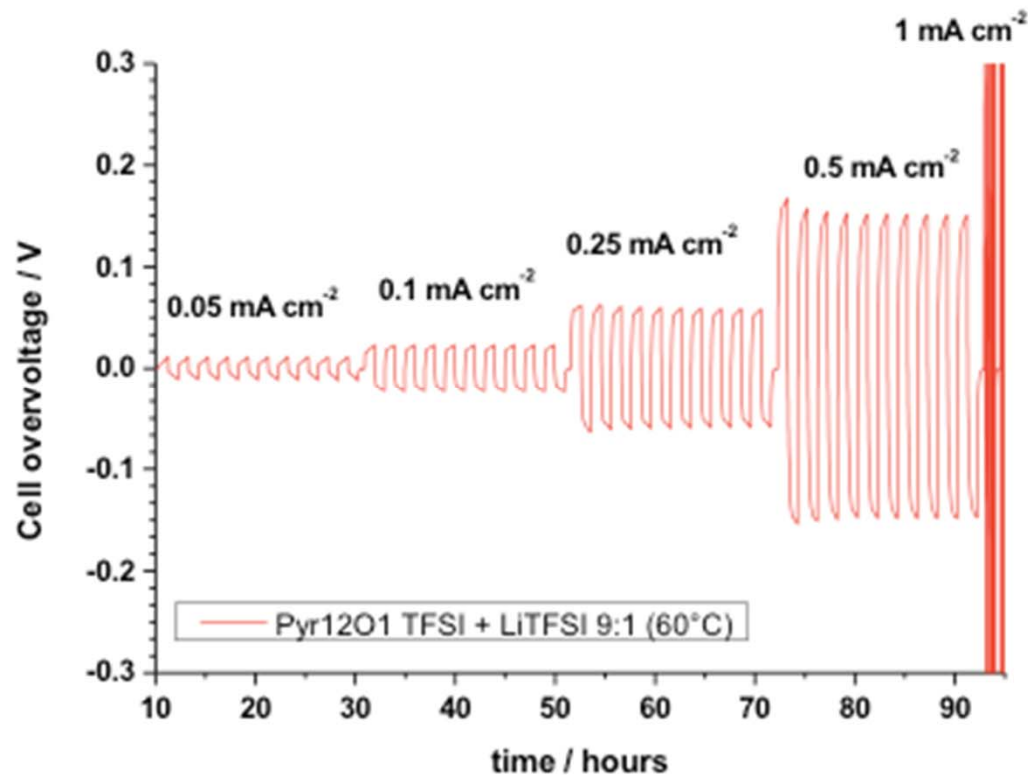
Li / 0.1LiTFSI-0.9PYR₁₄FSI/Ni (20° C, various current rates, sealed cell)



Limited rate performance

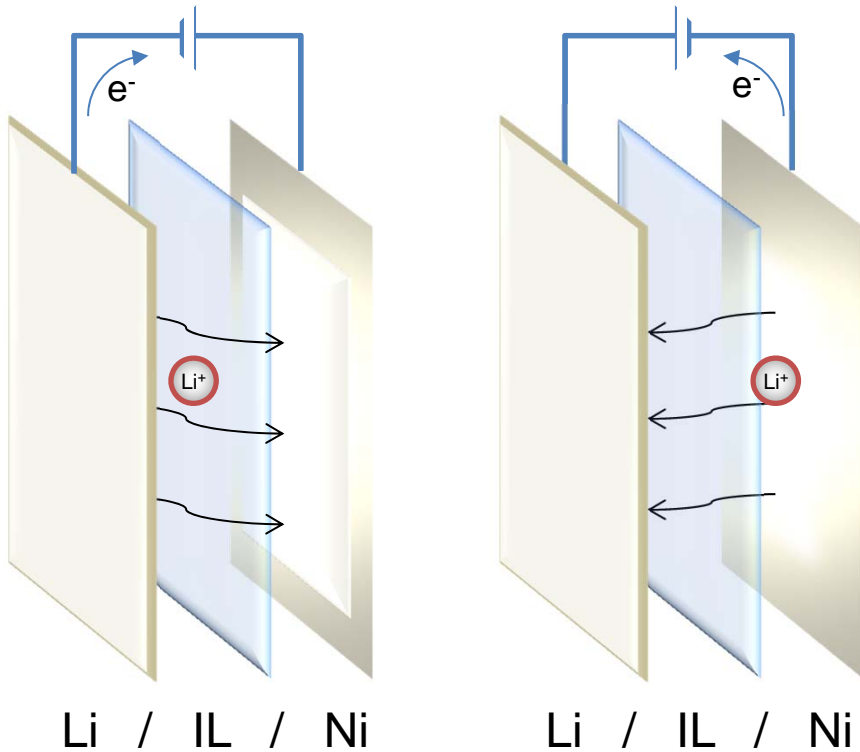
Li plating/stripping from IL-based electrolytes

Li / 0.1LiTFSI-0.9PYR₁₂O₁TFSI/Li (60° C sealed cell)



Open challenge: 1 mAcm⁻²

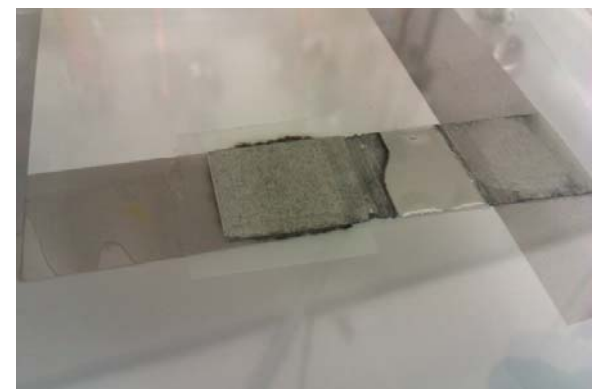
Effect of (dry) air on Lithium plating/stripping



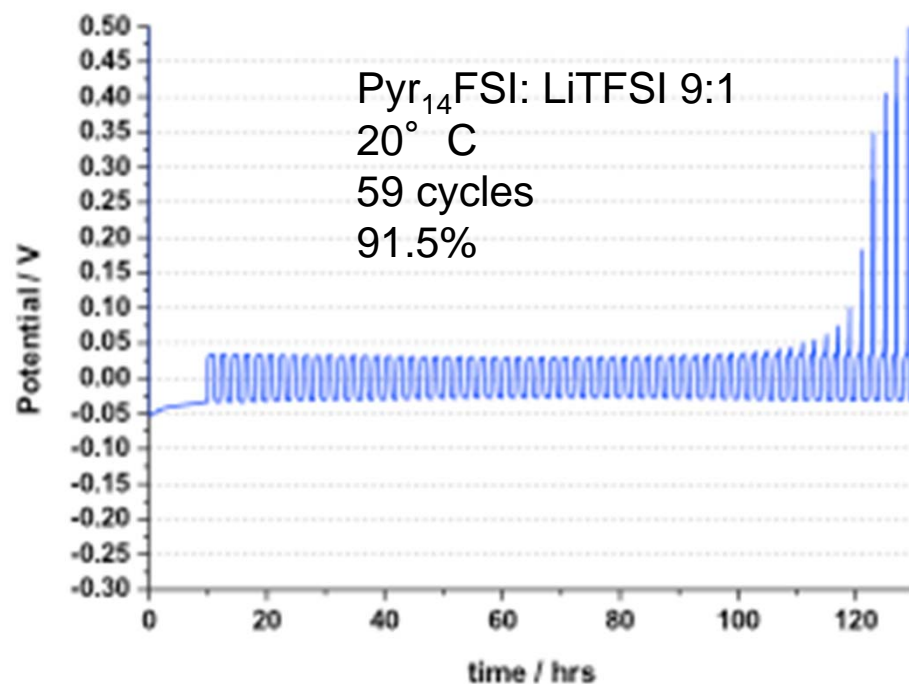
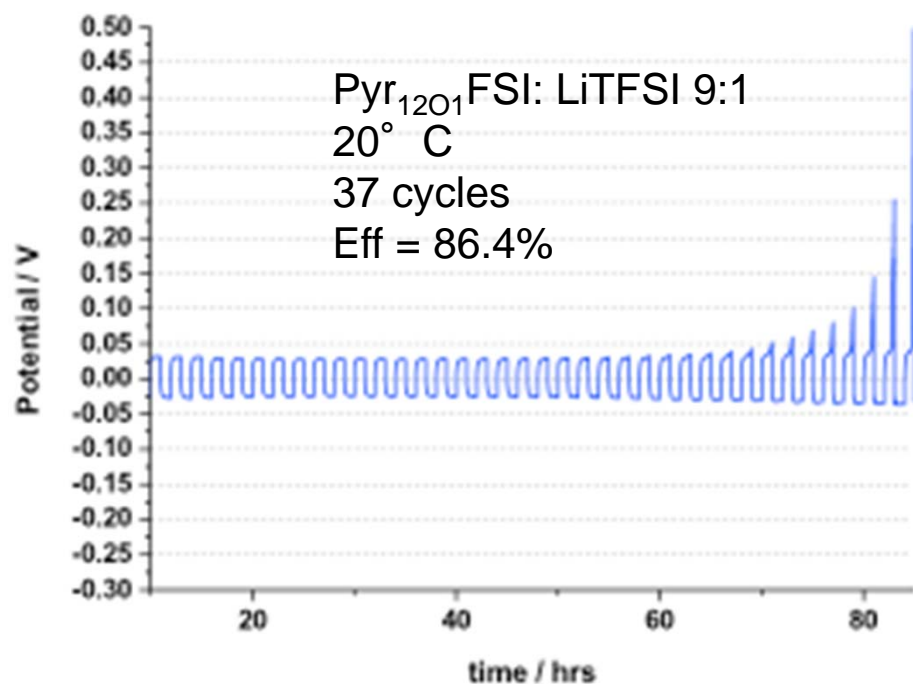
10h plating at 0.1 mA cm^{-2}

Cycles of 1h stripping / 1h plating
 0.1 mA cm^{-2} , $V_{\text{cut-off}} = 0.5\text{V}$

$$Eff = 1 - \frac{Q_p}{2nQ_c}$$



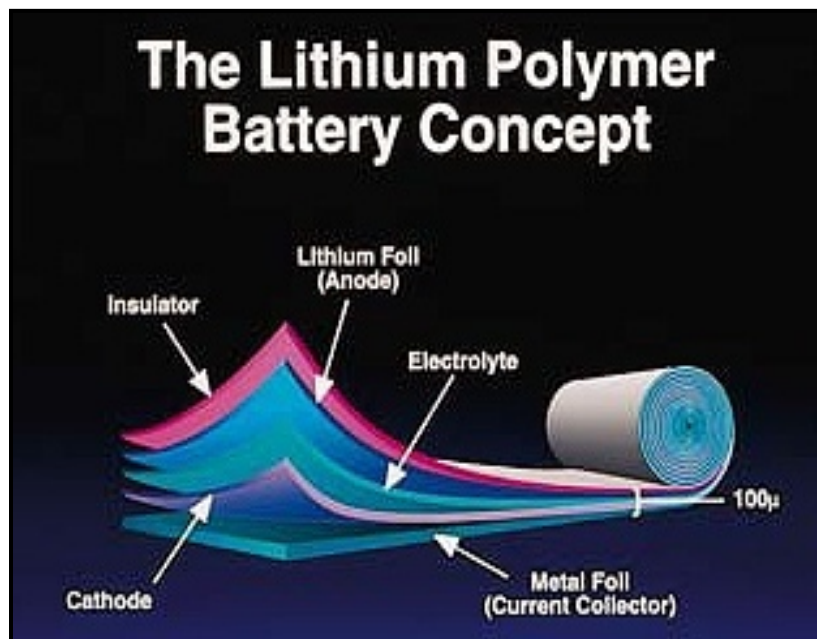
Pyr_{12O1} FSI: LiTFSI 9:1 (open dry air)



Good cycling performance even in presence of air

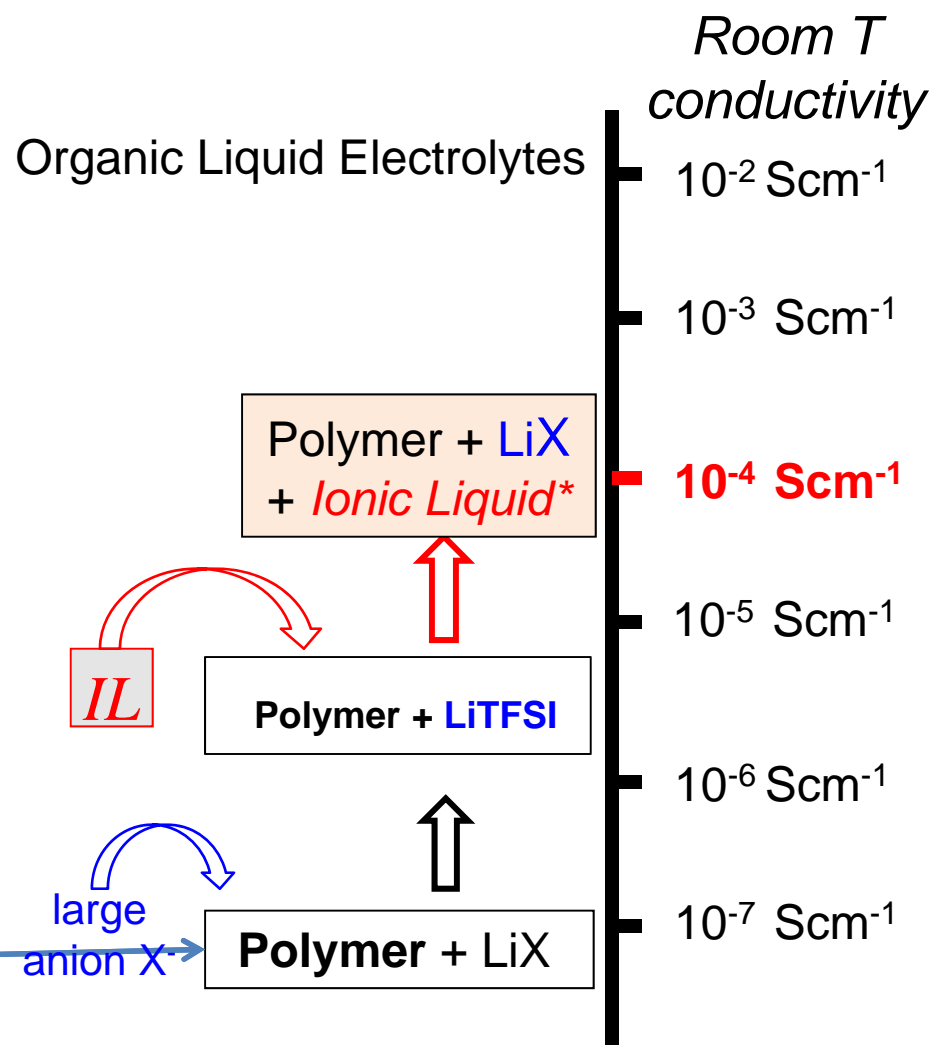
PYR₁₄ is more stable than PYR_{12O1} in dry air

Lithium Metal Polymer Batteries (LMPBs)



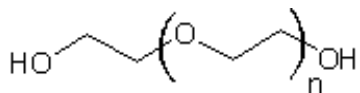
- no liquid compounds
- high safety
- high energy density
- very good processability ...

but *poor RT conductivity*

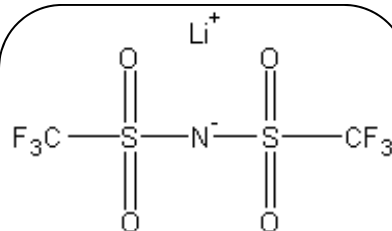


Incorporation of ILs has resulted in high ionic conductivity SPEs

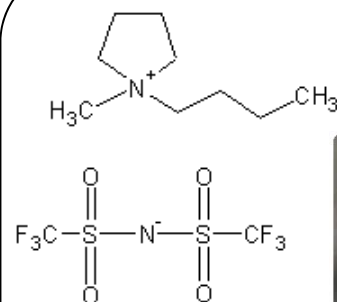
PEO-LiX-IL electrolytes: SPEs



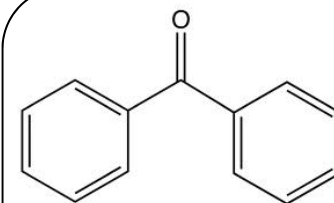
Poly(ethylene oxide)
PEO, $n \sim 91.000$
Molecular weight ~ 4 Mio.



Li TFSI



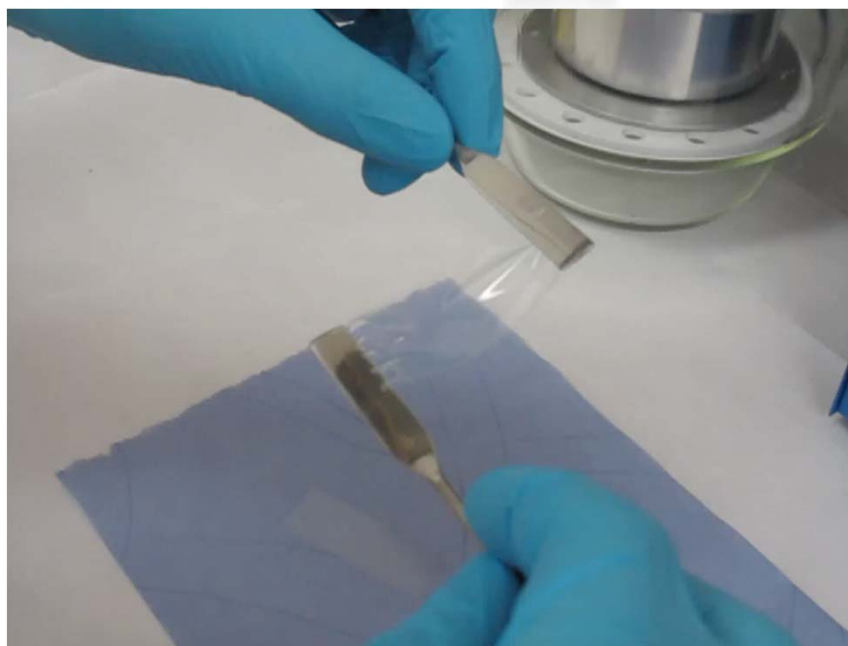
PYR₁₄TFSI



Benzophenone

Composition is given as:

Molar ratio of P(EO) : Li salt : IL
[20 : 2 : 4]

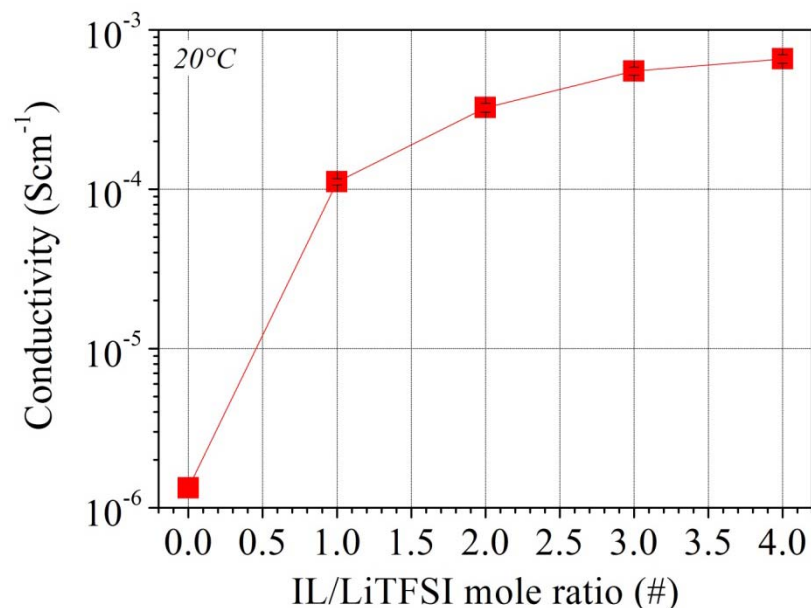


Good adhesiveness

Excellent mechanical properties

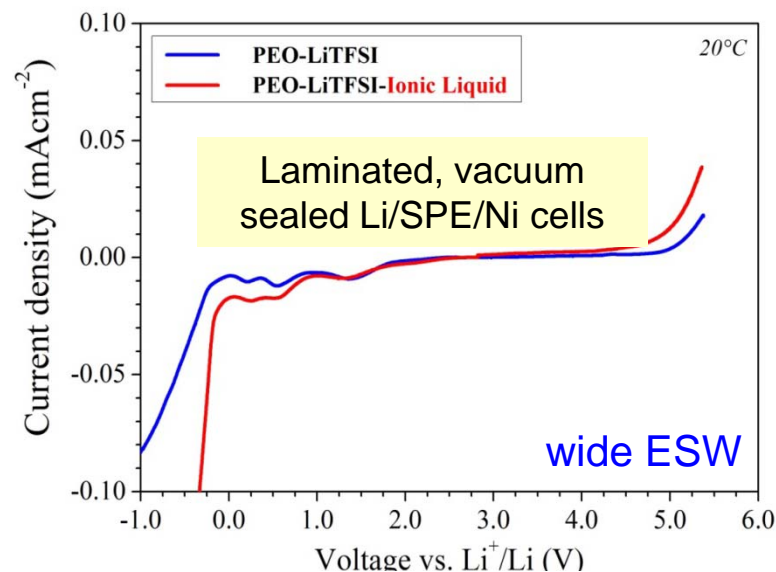
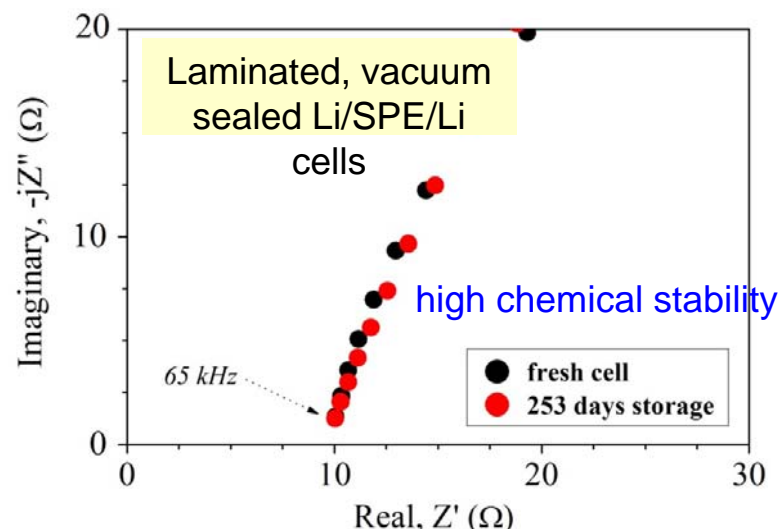
PEO-LiX-IL electrolytes: conductivity and stability

PEO : LiTFSI : $\text{PYR}_{14}\text{TFSI}$ = 20:2:4 (mol)



High ionic conductivity

Excellent mechanical, chemical and electrochemical stabilities

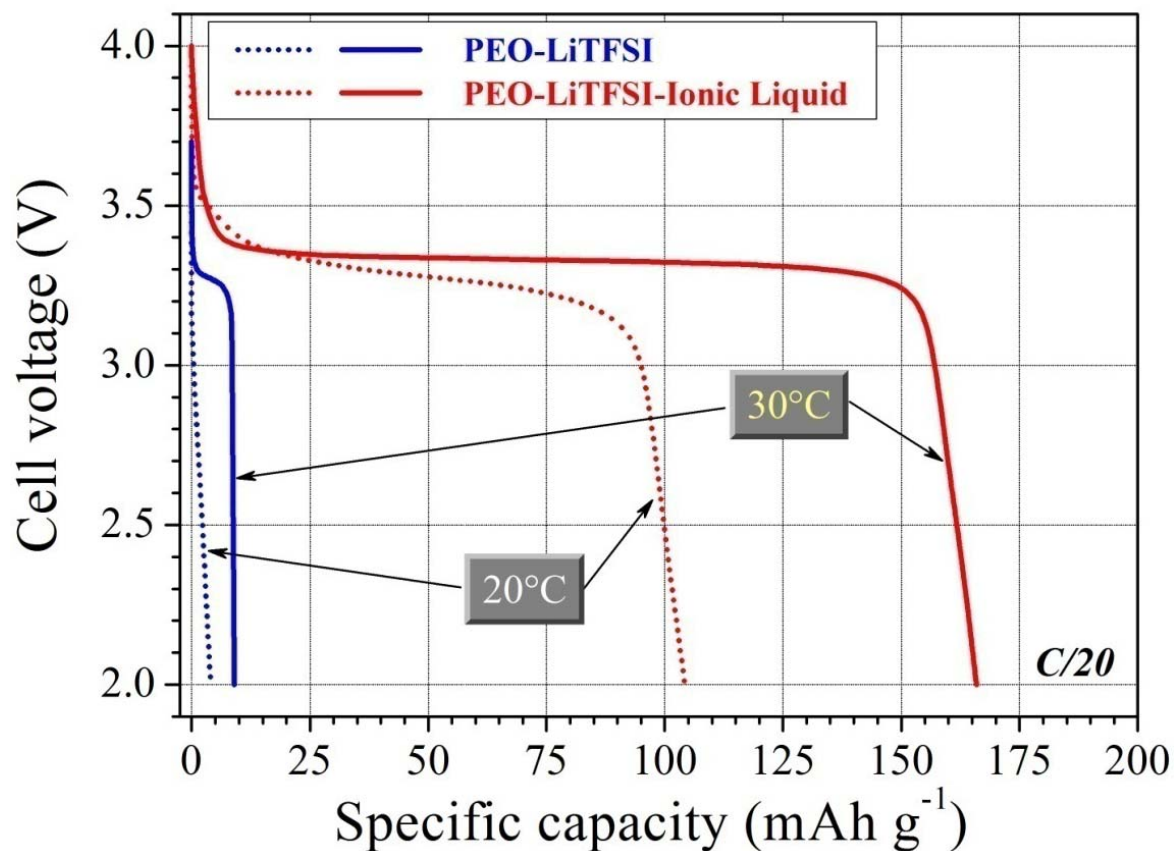


Lithium metal electrode operation appears feasible in IL-based electrolyte



Li/ IL-based electrolyte/ LFP
1Ah prototype

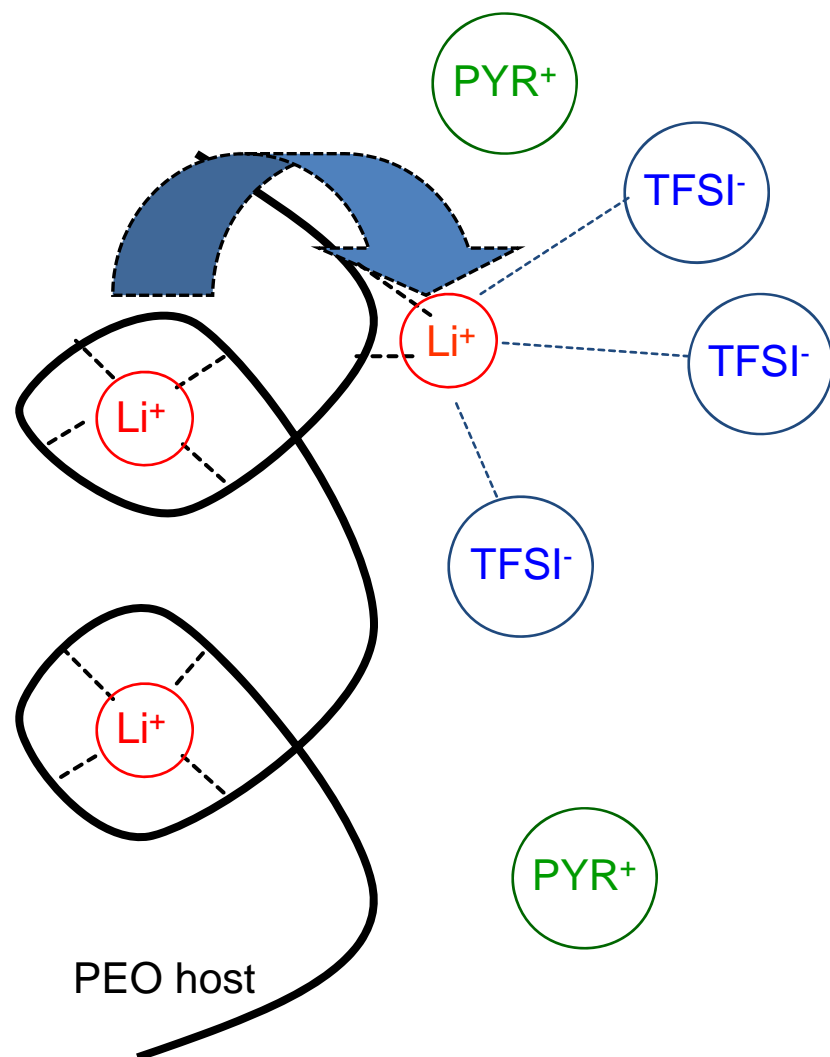
Battery tests: PEO-IL-LiX electrolyte (RT tests)



- Very large capacity increase due to IL incorporation > 100 mAhg⁻¹ at 20°C
- Theoretical capacity (170 mAhg⁻¹) at 30°C

PEO-LiX-IL electrolytes

How do ILs enhance Li^+ conduction?



PEO-LiX electrolyte: Li^+ is strongly coordinated by PEO host

PEO-LiX-IL electrolyte:

solid ternary system composed by a polymer host (PEO) and two salts (LiTFSI and $\text{PYR}_{14}\text{TFSI}$)

- very weak interactions between PYR_{14}^+ and TFSI^- [1]
- no interaction between PYR_{14}^+ and PEO host (Raman & NMR) [2,3]
- **strong interactions between Li^+ and TFSI^- (in excess)**

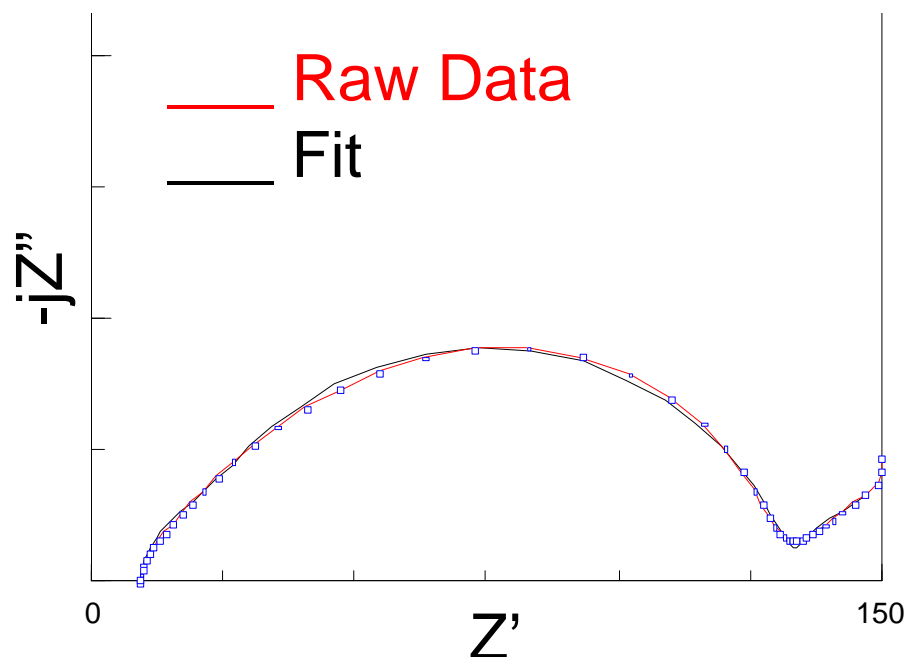
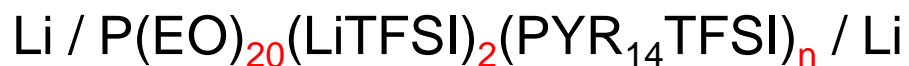
the strength of the Li --PEO coordination lowered
the lithium mobility (Li^+ conduction) is promoted

¹M. Castriota, T. Caruso, R. G. Agostino, E. Cazzanelli, W. A. Henderson, and S. Passerini, *J. Phys. Chem. A* **109** (2005) 92

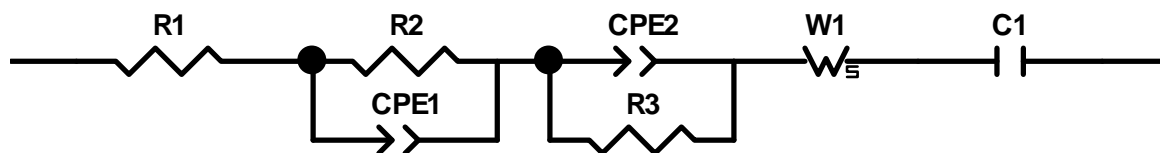
²I. Nicotera, C. Oliviero, W. A. Henderson, G. B. Appetecchi, and S. Passerini *J. Phys. Chem. B* **109** (2005) 22814.

M. Joost et al, *Electrochim. Acta*, in press 2012

PEO-LiX-IL electrolytes: Li interface (40° C)



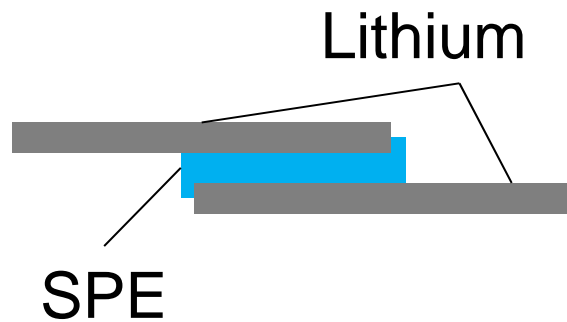
- R_1 = bulk electrolyte resistance
- R_2 = Li/SPE interface 1
- CPE_1 = Li/SPE interface 1
- R_3 = Li/SPE interface 2
- CPE_2 = Li/SPE interface 2
- W_1 = Warburg impedance
- C_1 = electrode limit capacitance



Impedance is localized at the Li/SPE interface

Transport properties by DC tests

How much current can be flown through the SPE?



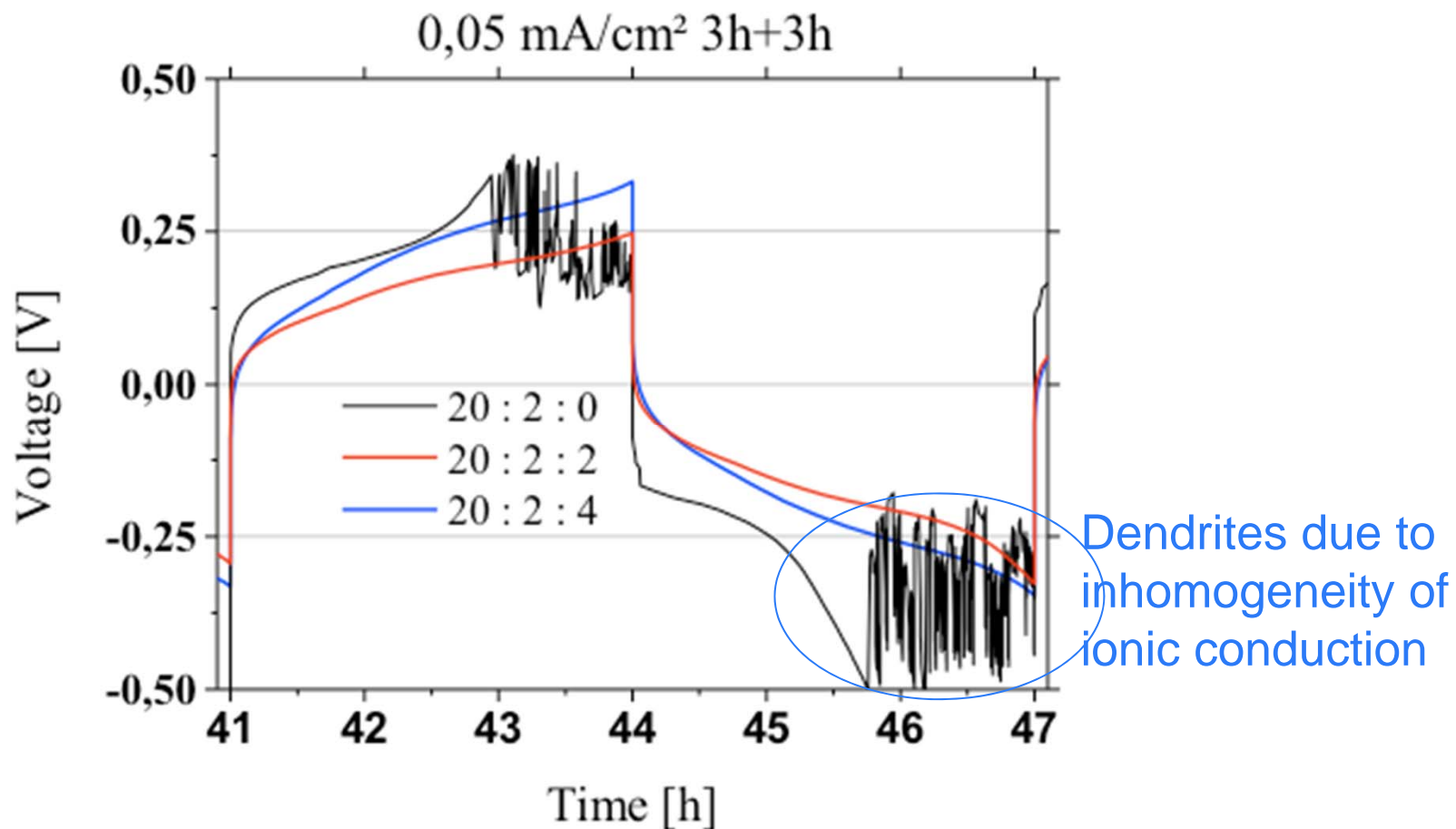
Galvanostatic and
Potentiostatic
experiments

Capacity of Li foil (Chemetall):

$$50 \mu\text{m} * 1 \text{ cm}^2 = 5\text{mm}^3 \rightarrow \mathbf{10.32 \text{ mAh/cm}^2}$$

Transport properties by galvanostatic tests (30° C)

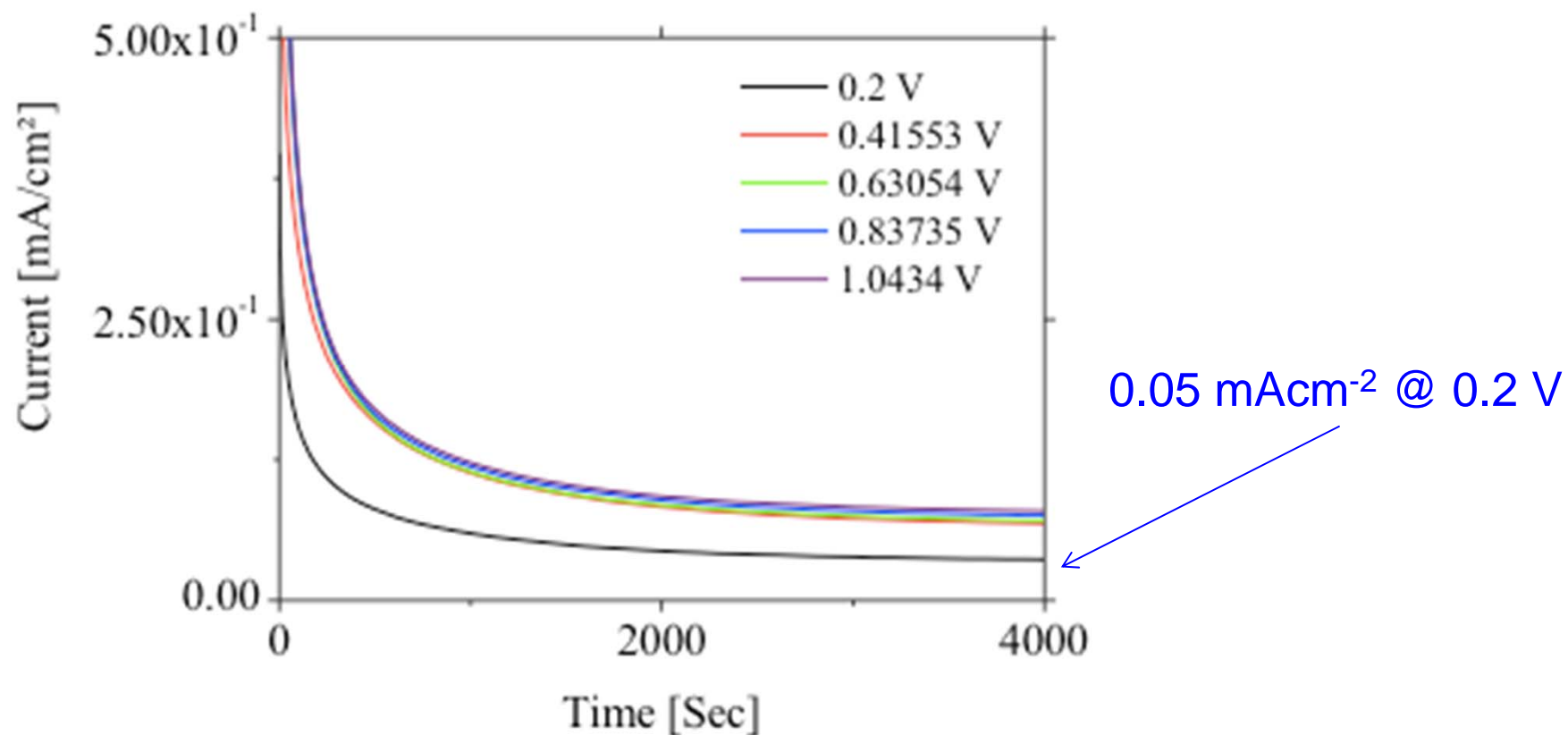
Li / P(EO)₂₀(LiTFSI)₂(PYR₁₄TFSI)_n / Li



IL containing electrolytes perform well at 30° C

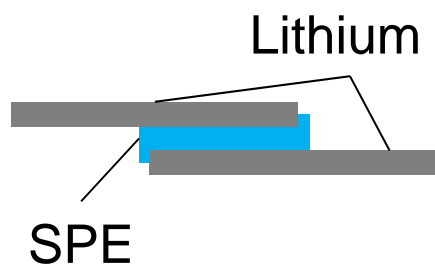
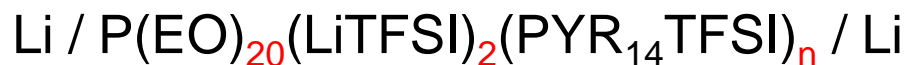
Transport properties by potentiostatic tests (30° C)

Li / P(EO)₂₀(LiTFSI)₂(PYR₁₄TFSI)₄ / Li

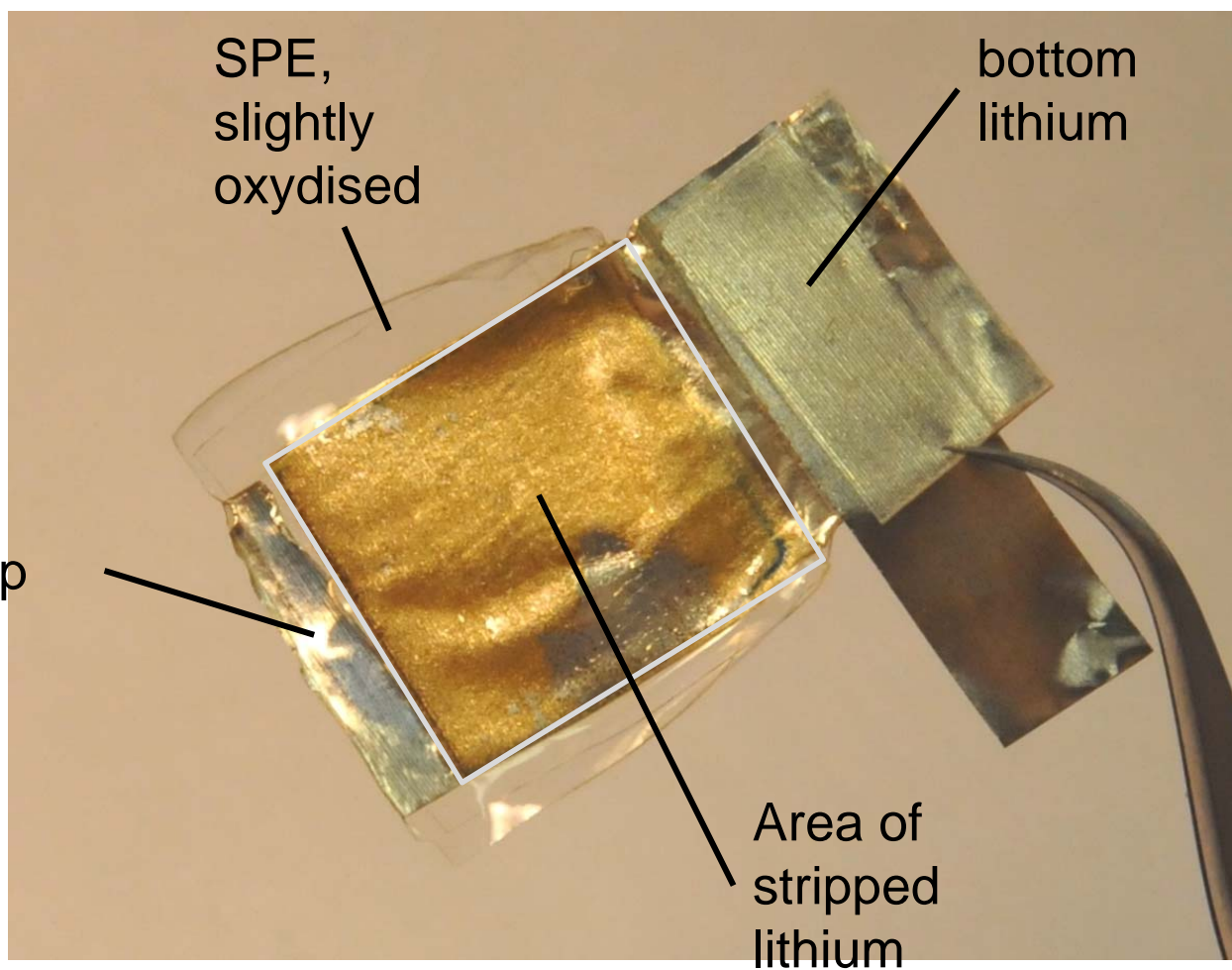


Good agreement between galvanostatic and potentiostatic results

Transport properties by DC tests

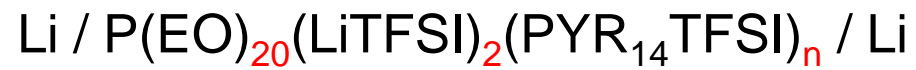


Left over
edge of top
lithium



Complete stripping of 50 μm of lithium (ca. 10 mAh cm^{-2})

Transport properties by DC tests



Open cell



Plated lithium

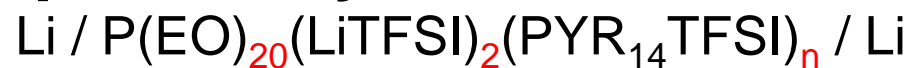


SPE

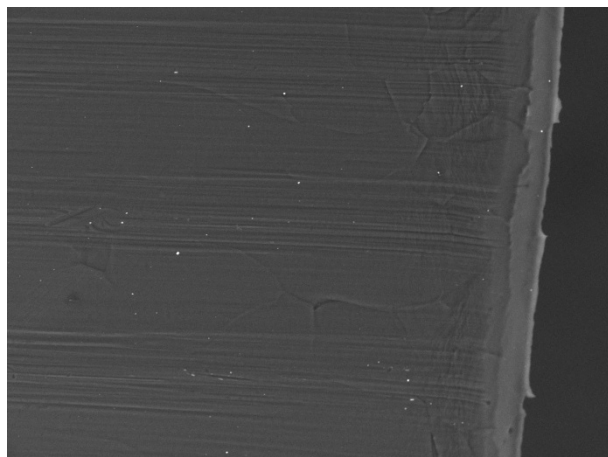


Opened cell after full lithium plating

Transport properties by DC tests

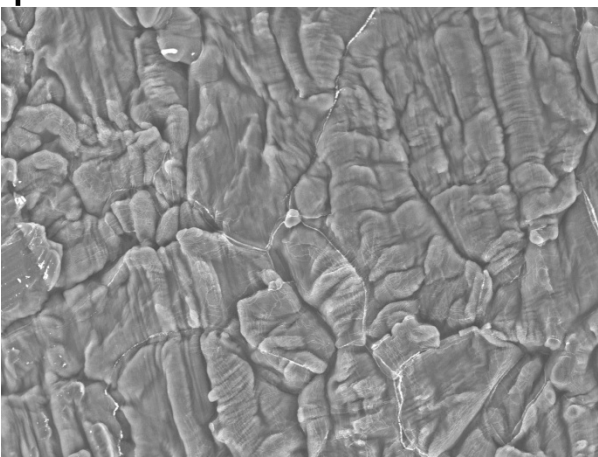


fresh lithium



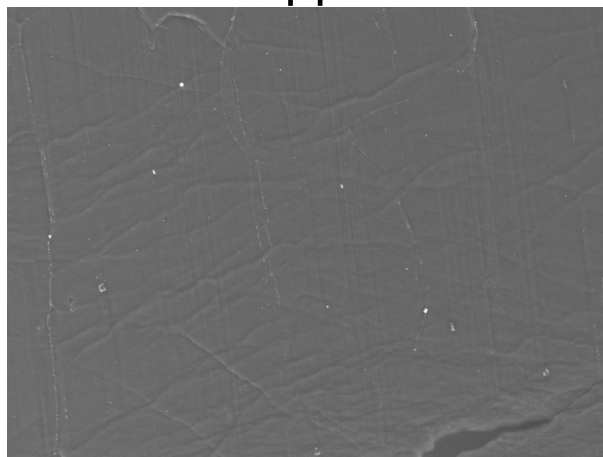
fresh_Li0008 2012.04.27 14:55 D2.1 x200 500 um

plated lithium

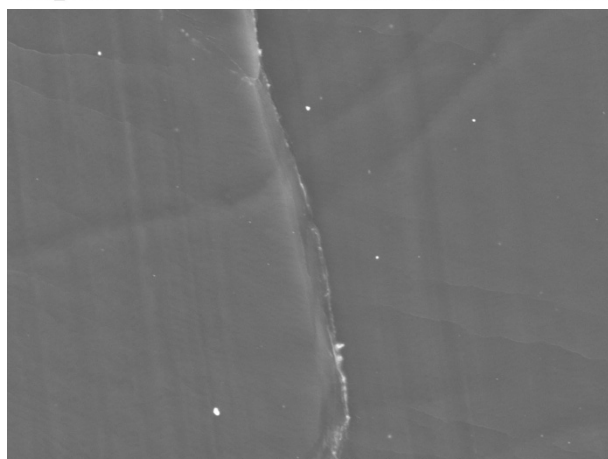


plated_Li0007 2012.04.26 17:07 D2.1 x200 500 um

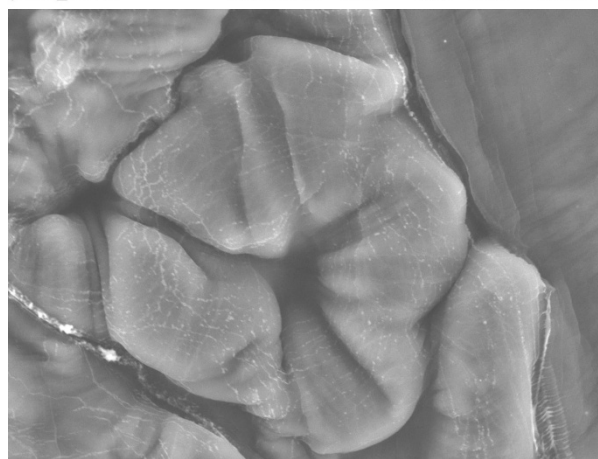
SPE on stripped lithium



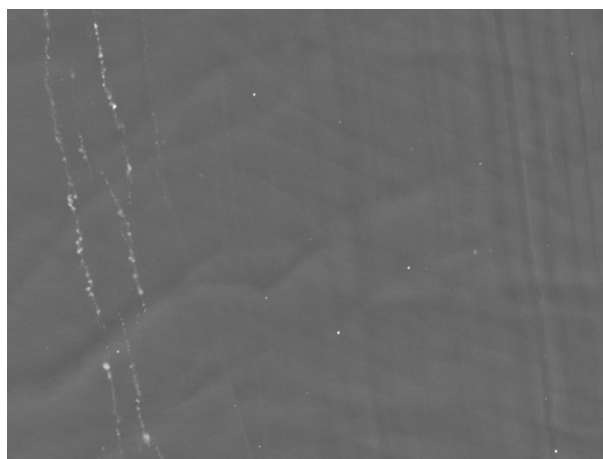
Li0010 2012.05.16 10:27 D2.0 x200 500 um



Li0005 2012.05.16 10:17 D2.0 x1.0k 100 um
stripped



plated_Li0005 2012.04.26 17:02 D2.1 x1.0k 100 um
full_strip_SPE



Li0008 2012.05.16 10:25 D2.0 x1.0k 100 um
stripped

SEM of fresh lithium, plated lithium and SPE

Conclusions

Ionic Liquid mixtures with lithium salts as such and in presence of polymers show very promising performance as Li-ion conductors.

IL-based electrolytes:

- easy to recycle (low volatility)
- wide storage temperature range
- low flammability components
- high temperature operation capability ($>60^{\circ}\text{C}$)

IL-based electrolytes appear promising for safe, high energy and moderate power applications (such as full EV and stationary storage).

Acknowledgements

German and NRW funding agencies

The European Commission for the financial support of the projects:

ILLIBATT (Ionic Liquid based **L**ithium **BAT**teries)

LABOHR (**L**ithium – **A**ir **B**atteries with split **O**xygen **H**arvesting and **R**eduction processes)

GREEN LION (Advanced Manufacturing Processes for Low Cost **GREENer Li-ION** Batteries)

All members of WWU-MEET and ENEA battery groups

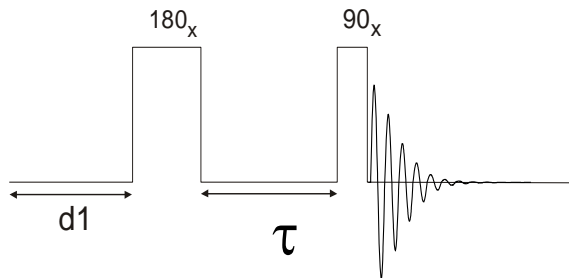
Back up

Methods

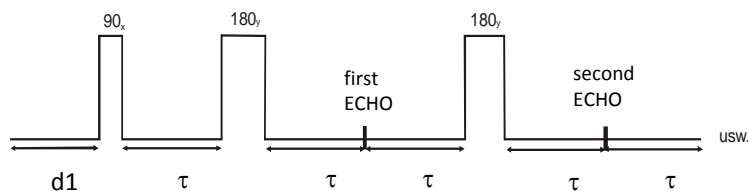
local motion

relaxation measurements

- 1) spin-lattice-relaxation time T_1 :
inversion recovery



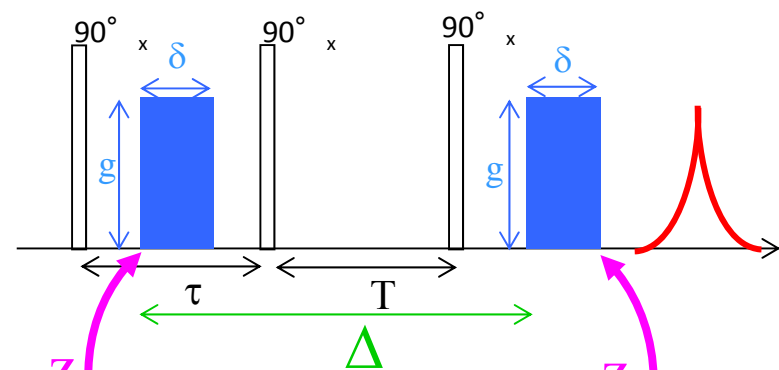
- 2) spin-spin-relaxation time T_2 :
CPMG-sequence



macroscopic motion

diffusion measurements:

pulsed gradient stimulated echo sequence



z_1
'mark' a spin
position by the
precession phase

z_2
'read' the spin
position after
time Δ

g : gradient strength

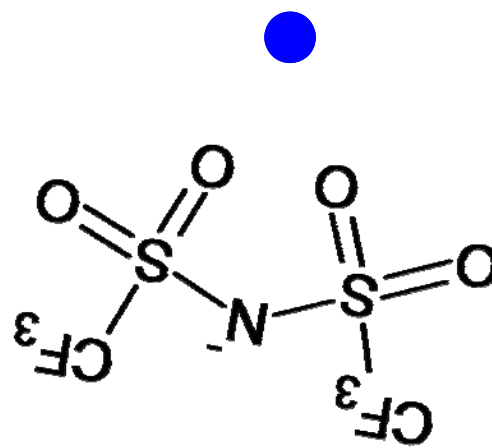
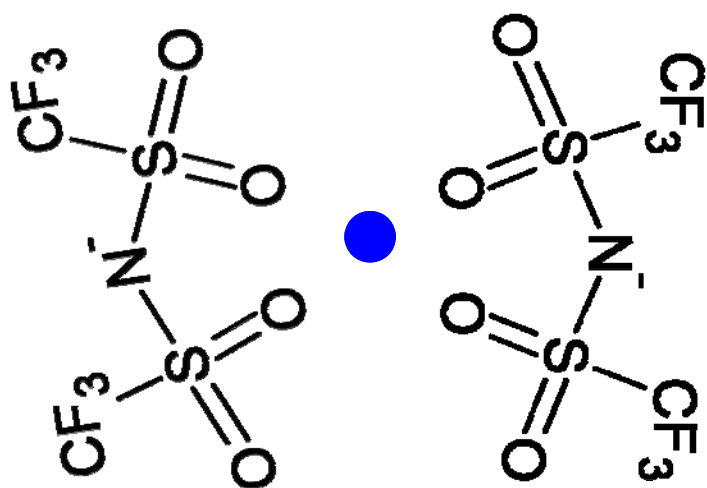
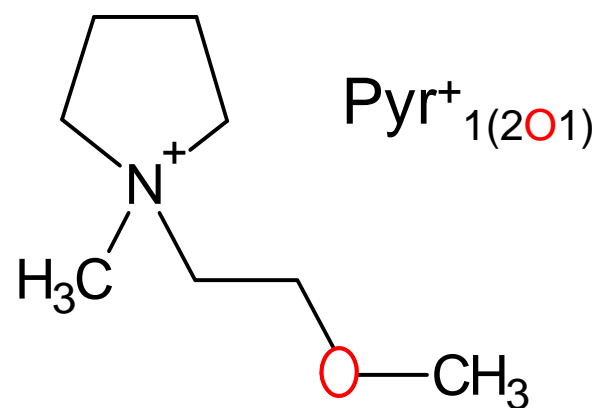
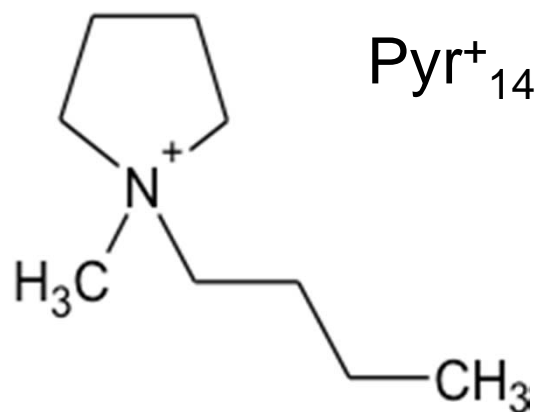
δ : gradient duration

Δ : diffusion time

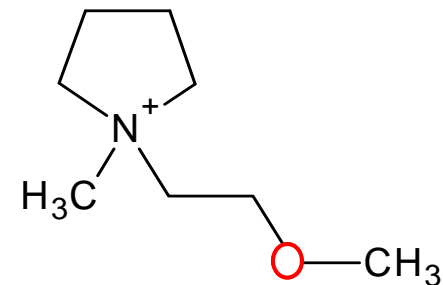
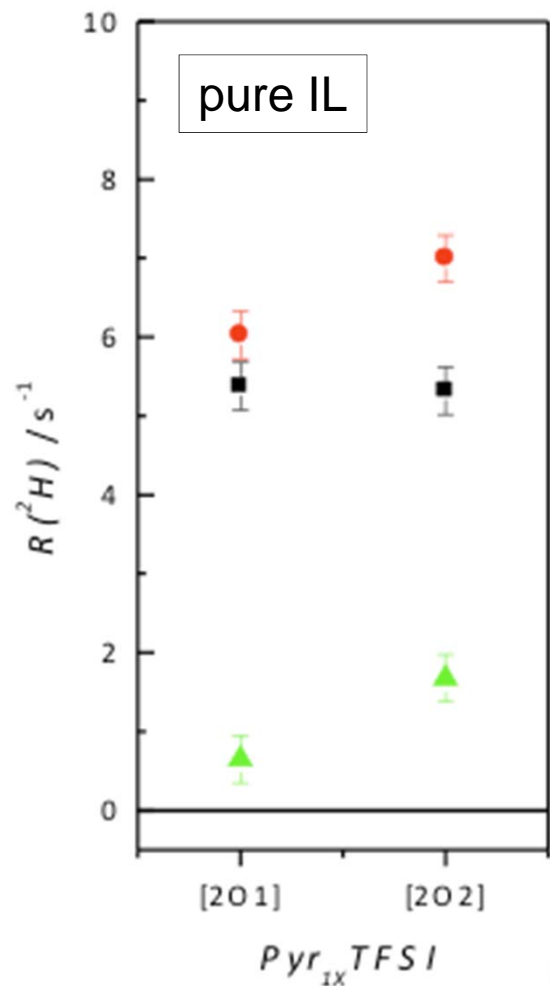
τ : evolution time

z_1 and z_2 : spin position before and after the gradients

Is the Li^+ --- TFSI^- ion-pairing avoidable?



Cation aggregation in ILs with polar side chains



R_2 dependent on chain length in pure systems

⇒ slow local motions influenced

for systems with Li salt ΔR does

R_1 independent on chain length in pure systems

⇒ fast local motions not influenced

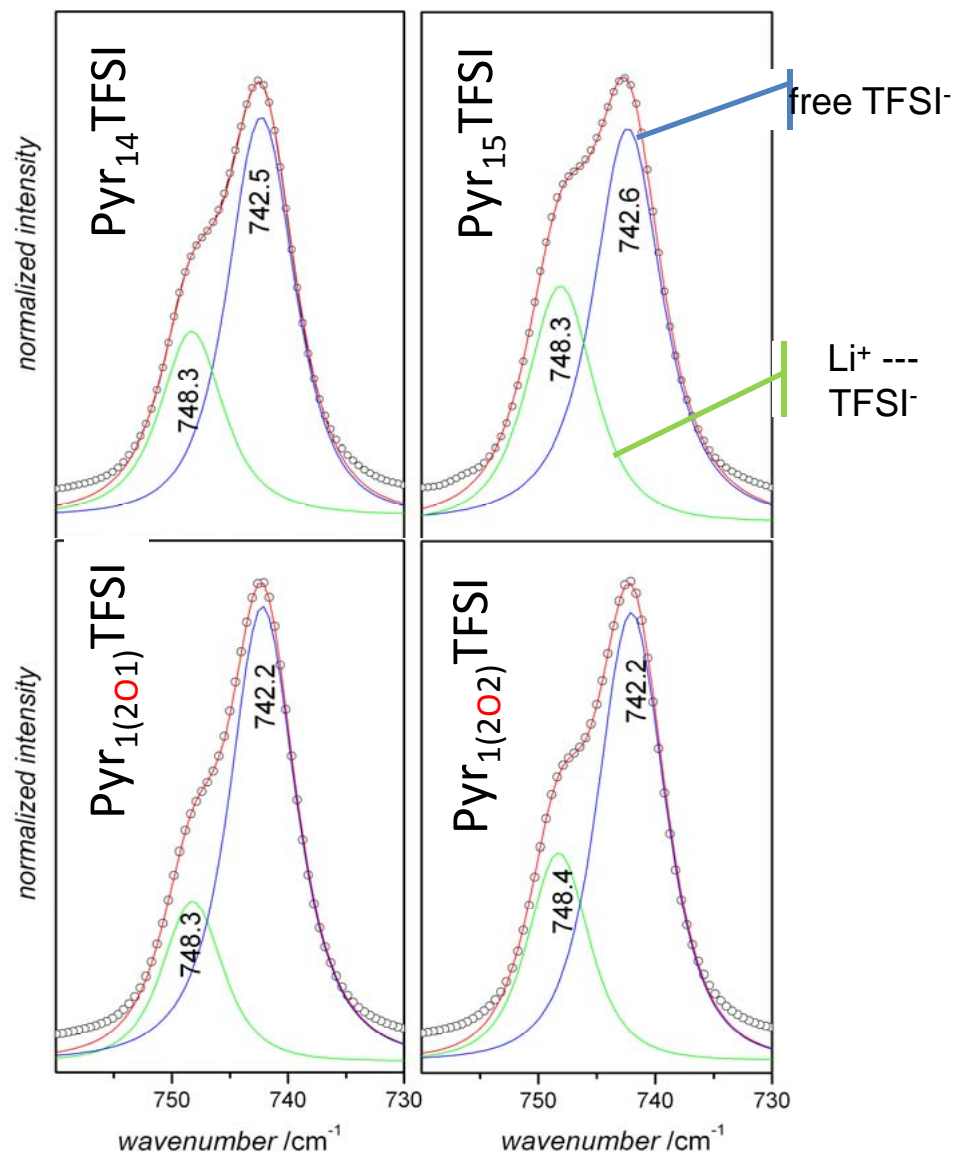
ΔR changes with chain length

⇒ anisotropic arrangement

⇒ aggregation of cations w/ polar group

⇒ cations do not aggregate when a lithium salt is dissolved

PYR⁺_{1A} – Li⁺ – TFSI⁻ interactions

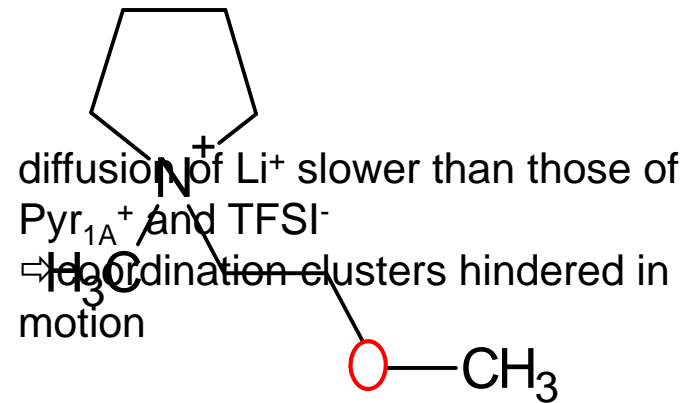
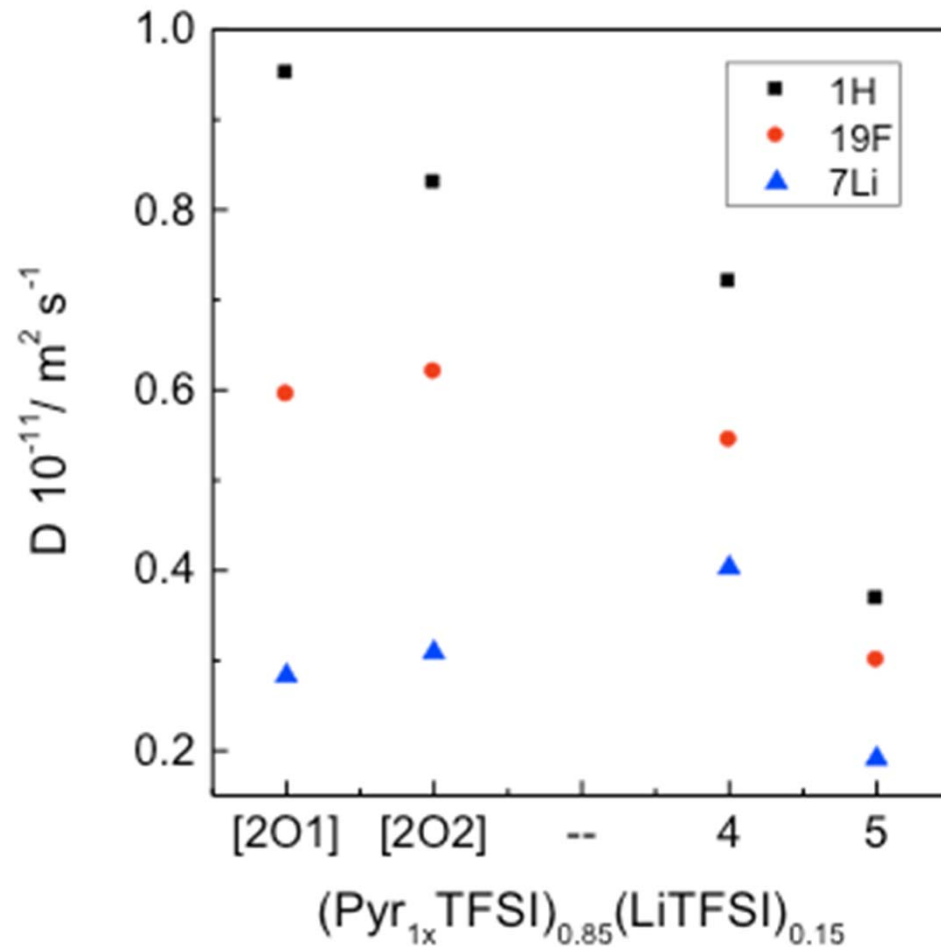


Raman

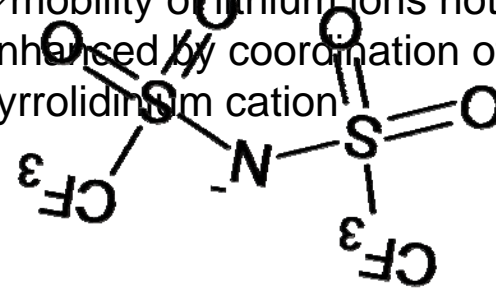
	n in Li ⁺ -(TFSI) _n
(Pyr ₁₄ TFSI) _{0.85} (LiTFSI) _{0.15}	2.12
(Pyr ₁₅ TFSI) _{0.85} (LiTFSI) _{0.15}	2.34
(Pyr ₁₍₂₀₁₎ TFSI) _{0.85} (LiTFSI) _{0.15}	1.55
(Pyr ₁₍₂₀₂₎ TFSI) _{0.85} (LiTFSI) _{0.15}	1.87

Addition of oxygen into side chain reduces the number of TFSI anions in cluster.

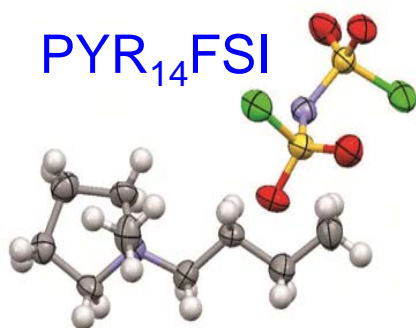
Diffusion coefficient



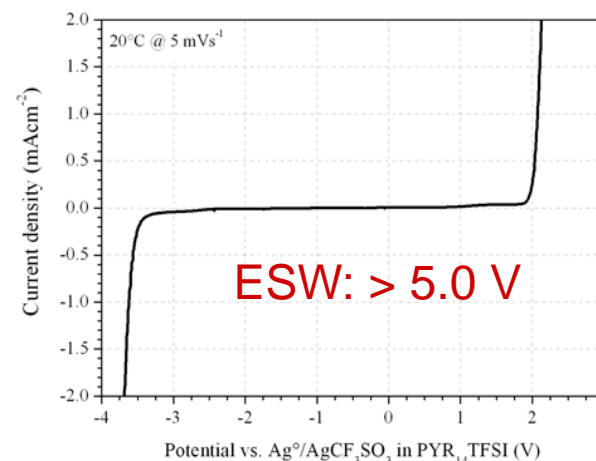
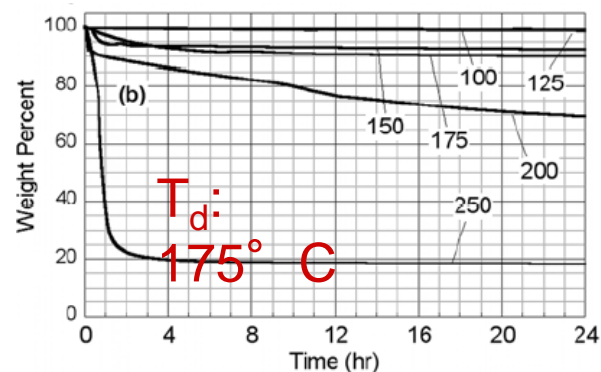
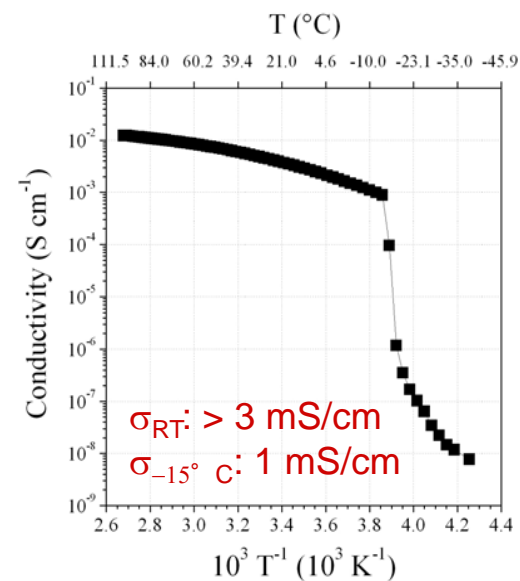
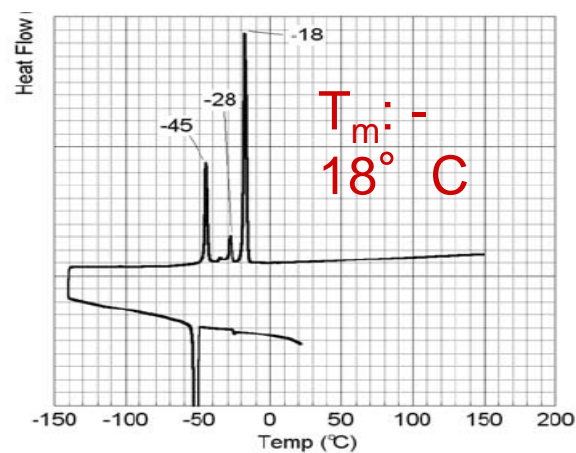
difference of diffusion coefficients larger for polar side chain than for non-polar side chain
 \Rightarrow mobility of lithium ions not enhanced by coordination of pyrrolidinium cation



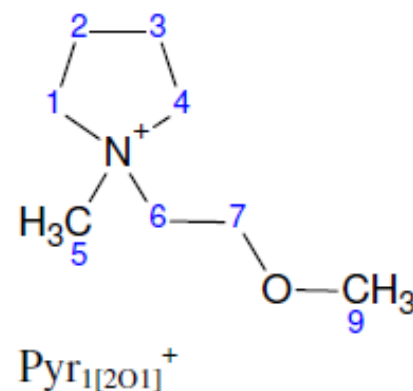
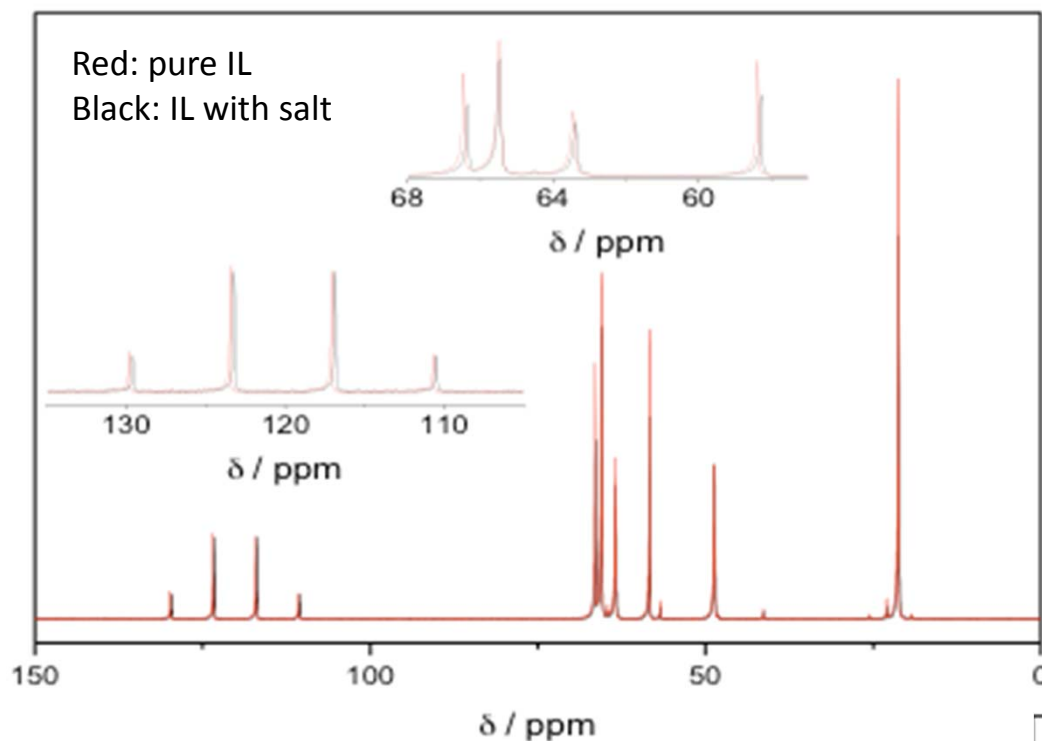
Tailored Ionic Liquid for Lithium Batteries



Purity > 99.5%
H₂O < 1ppm



Pyr⁺₁₄ – Li⁺ – TFSI⁻ interactions



NMR-chemical shift

NMR features of oxygen
neighbours are strongly affected as
a result of the Li --- IL Cation
interactions

	pure	+ LiTFSI
δ / ppm	assignment ($J^2(\text{CH}) = 2.5\text{Hz}$)	$\Delta\delta$ / ppm (Hz)
58.377	C ⁹ (quadruplet)	-0.09 (-4.7)
63.447	C ⁶ (triplet)	-0.07 (-3.4)
65.479	C ¹ , C ⁴ (triplet)	-0.00 (-0.0)
66.454	C ⁷ (triplet)	-0.09 (-4.7)
120.271	-CF ₃ (quadruplet, $J^2(\text{CF})$ =80.15Hz)	-0.23 (-11.7)

ILLIBATT Project achievements

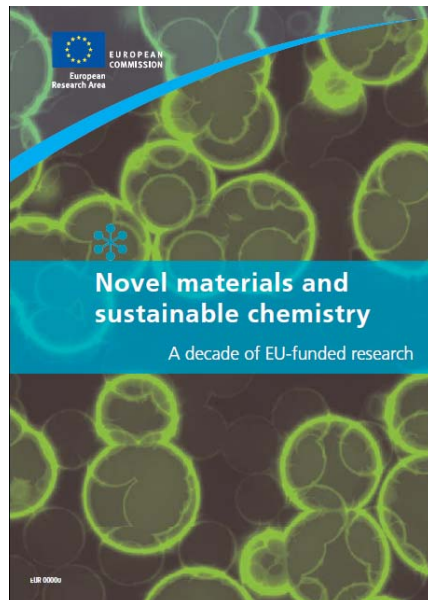
Demonstrated feasibility of Ionic Liquid-based batteries at the pre-industrial scale

ILLIBATT has been selected together with other most successful projects within the FP5 and FP6 calls

“Materials for a sustainable chemistry”

European Commission, Research Directorate-General

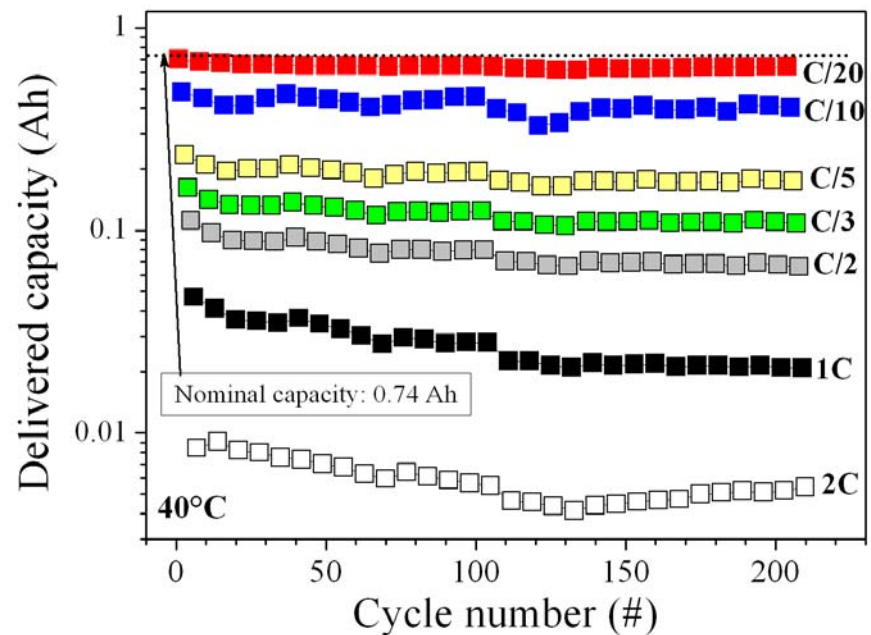
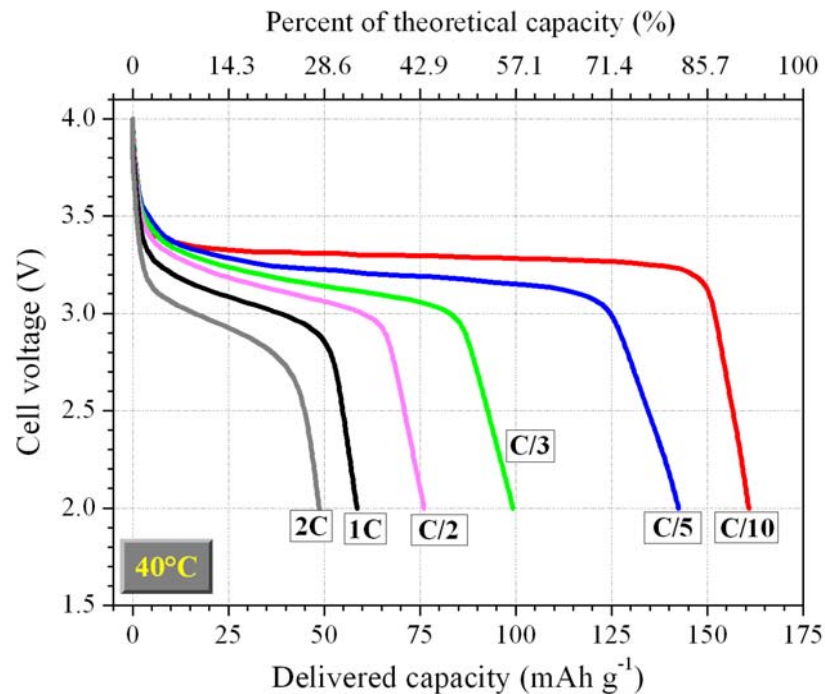
Directorate G-Industrial Technologies-Value-added Materials



ILLIBATT Cells:

- Do not contain volatile components
- Halides are present only in the easy-to-recycle electrolyte
- All binders are halide-free

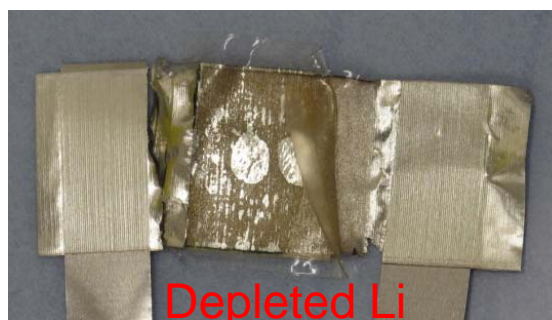
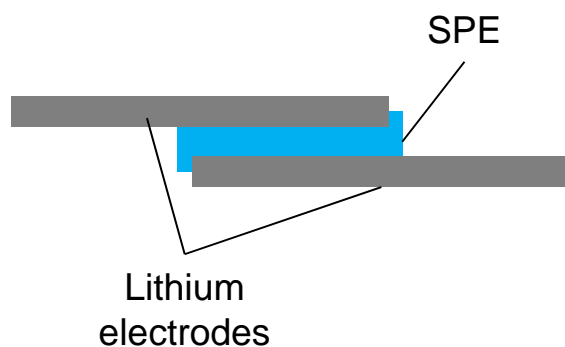
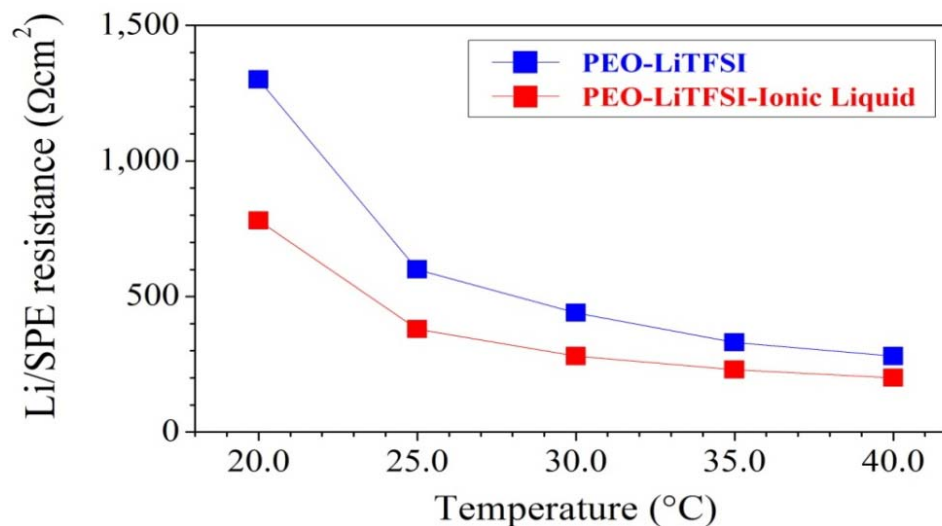
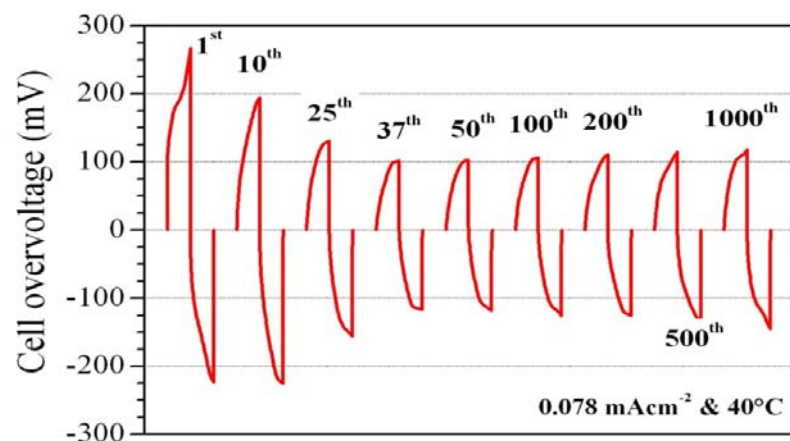
Battery tests: PEO-IL-LiX electrolyte (near RT tests)



Large capacities (140 mAhg⁻¹) up to medium rates (C/5)
Moderate capacities (50-75 mAhg⁻¹) even at high rates (2C-1C)
> 90% of initial capacity is delivered upon about 200 cycles

PEO-LiX-IL electrolytes: Li interface

PEO : LiTFSI : $\text{PYR}_{14}\text{TFSI}$ = 20:2:4 (mol)



Excellent lithium plating/stripping performance
Lowered interface resistance